

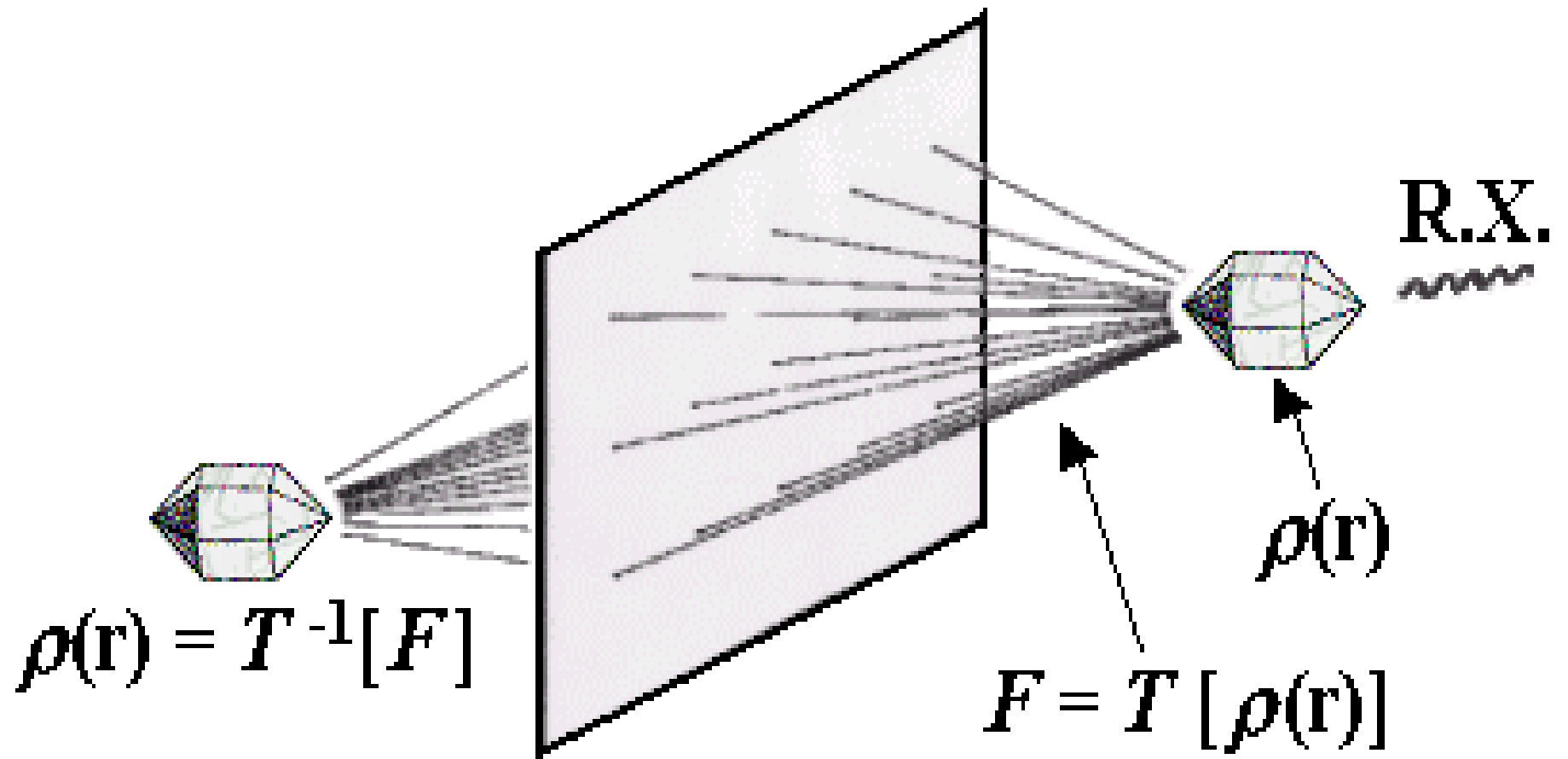
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Direct Methods

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Let us answer the following questions:

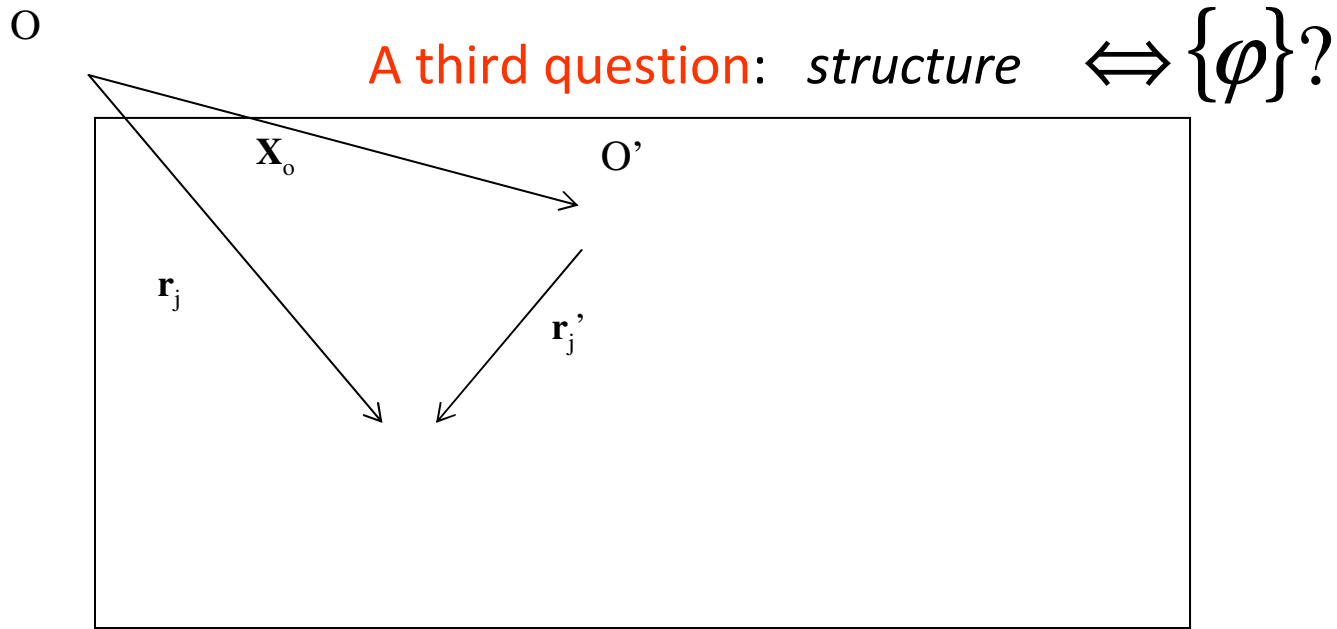
crystal structure $\Rightarrow \{ |F|^2 \}$?

$\{ |F|^2 \} \Rightarrow$ crystal structure ?

$$\begin{aligned} |F_h|^2 &= \sum_{j=1}^N f_j \exp(2\pi i h r_j) \sum_{j=1}^N f_j \exp(-2\pi i h r_j) \\ &= \sum_{i,j=1}^N f_i f_j \exp[2\pi i h (r_i - r_j)] \\ &= \sum_{j=1}^N f_j^2 + \sum_{i \neq j=1}^N f_i f_j \exp[2\pi i h (r_i - r_j)] \end{aligned}$$

As a consequence :

$$\{ |F|^2 \} \Leftrightarrow \rho(\mathbf{r})$$



$$\begin{aligned}
 F_h &= \sum_{j=1}^N f_j \exp(2\pi i h \mathbf{r}_j) = \sum_{j=1}^N f_j \exp(2\pi i h (\mathbf{X}_0 + \mathbf{r}'_j)) \\
 &= \exp(2\pi i h \mathbf{X}_0) \sum_{j=1}^N f_j \exp(2\pi i h \mathbf{r}'_j) \\
 &= \exp(2\pi i h \mathbf{X}_0) F'_h \\
 \rightarrow F'_h &= F_h \exp(-2\pi i h \mathbf{X}_0) = |F_h| \exp i(\varphi_h - 2\pi h \mathbf{X}_0)
 \end{aligned}$$

A fourth basic question

How can we derive the phases from the diffraction moduli ? This seems contradictory: indeed

The phase values depend on the origin chosen by the user, the moduli are independent of the user .

The moduli are ***structure invariants*** ,
the phases are not structure invariants.

Evidently, from the moduli we can derive information only on those combinations of phases (if they exist) which are structure invariants.

The simplest invariant : *the triplet invariant*

Use the relation

$$F'_h = F_h \exp (-2\pi i \mathbf{h} \mathbf{X}_0)$$

to check that the invariant $F_h F_k F_{-h-k}$ does not depend on the origin.

$$\begin{aligned} F'_h F'_k F'_{-h-k} &= |F_h| \exp i(\phi_h - 2\pi \mathbf{h} \mathbf{X}_0) |F_k| \exp i(\phi_k - 2\pi \mathbf{k} \mathbf{X}_0) \\ &\quad |F_{-h-k}| \exp i[\phi_{-h-k} + 2\pi(\mathbf{h} + \mathbf{k}) \mathbf{X}_0] \\ &= |F_h| |F_k| |F_{-h-k}| \exp i(\phi_h + \phi_k + \phi_{-h-k}) \end{aligned}$$

The sum $(\phi_h + \phi_k + \phi_{-h-k})$ is called ***triplet phase invariant*** .

Structure invariants

Any invariant satisfies the condition that the sum of the indices is zero:

$$\textit{doublet invariant} : F_h F_{-h} = |F_h|^2$$

$$\textit{triplet invariant} : F_h F_k F_{-h-k}$$

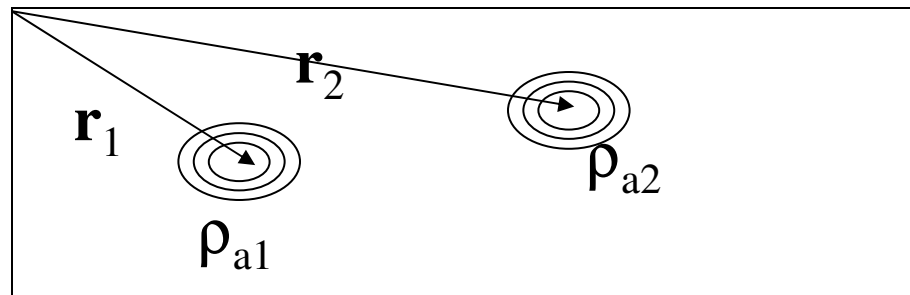
$$\textit{quartet invariant} : F_h F_k F_l F_{-h-k-l}$$

$$\textit{quintet invariant} : F_h F_k F_l F_m F_{-h-k-l-m}$$

.....

The prior information we can use for deriving the phase estimates may be so summarised:

1) **atomicity**: the electron density is concentrated in atoms:



$$\rho(\mathbf{r}) = \sum_{j=1}^N \rho_{aj}(\mathbf{r} - \mathbf{r}_j)$$

2) **positivity** of the electron density:

$$\rho(\mathbf{r}) > 0 \Rightarrow f > 0$$

3) **uniform distribution of the atoms in the unit cell.**

The Wilson statistics

- Under the above conditions **Wilson** (1942,1949) derived the structure factor statistics. The main results where:

- $$\langle |F_{\mathbf{h}}|^2 \rangle = \sum_{j=1}^N f_j^2 \quad (1)$$

- Eq.(1) is :

- a) resolution dependent (f_j varies with θ),

- b) temperature dependent: $f_j = f_j^0 \exp(-B_j \sin^2 \theta / \lambda^2)$

- From eq.(1) the concept of normalized structure factor arises:

$$E_{\mathbf{h}} = F_{\mathbf{h}} / (\sum_{j=1}^N f_j^2)^{1/2}$$

The Wilson Statistics

- $|E|$ -distributions:

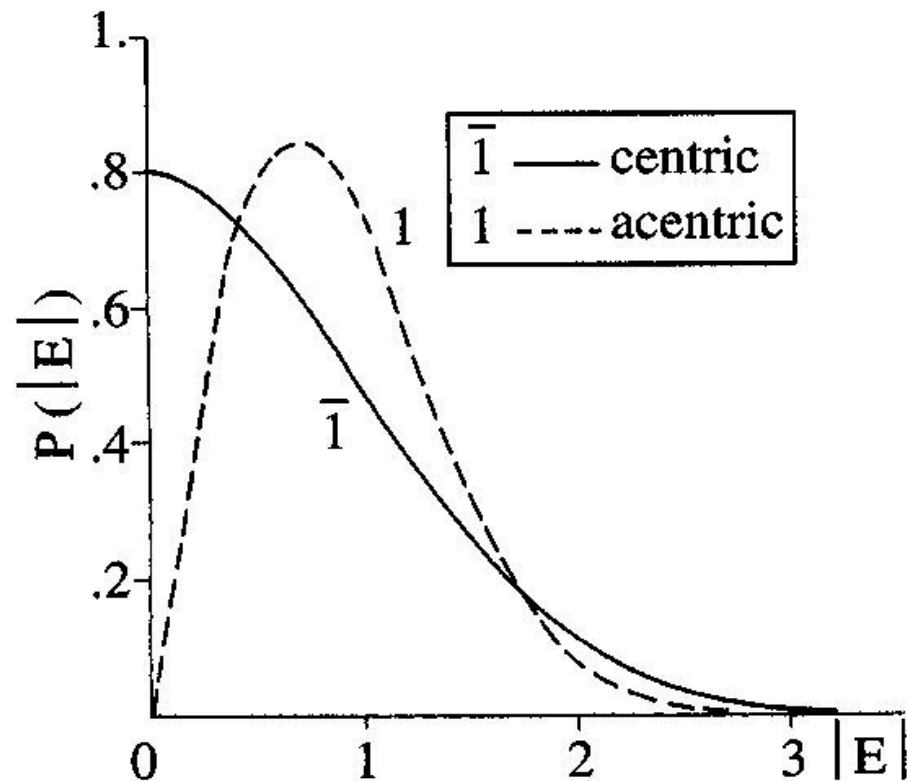
$$P_1(|E|) = 2|E| \exp(-|E|^2)$$

$$P_1(|E|) = \sqrt{\frac{2}{\pi}} \exp(-|E|^2 / 2)$$

and $\langle |E|^2 \rangle = 1$

in both the cases.

The statistics may be used to evaluate the average themel factor and the absolute scale factor.



The Wilson plot

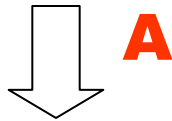
$$F_{\mathbf{h}} = \sum_{j=1}^N f_j \exp 2\pi i \mathbf{h} \mathbf{r}_j = \sum f_j^0 \exp \left(-B_j \frac{\sin^2 \theta}{\lambda^2} \right) \exp 2\pi i \mathbf{h} \mathbf{r}_j$$

A \rightarrow

$$\exp \left(-B \frac{\sin^2 \theta}{\lambda^2} \right) \underbrace{\sum f_j^0 \exp 2\pi i \mathbf{h} \mathbf{r}_j}_{F_{\mathbf{h}}^0}$$

s^2

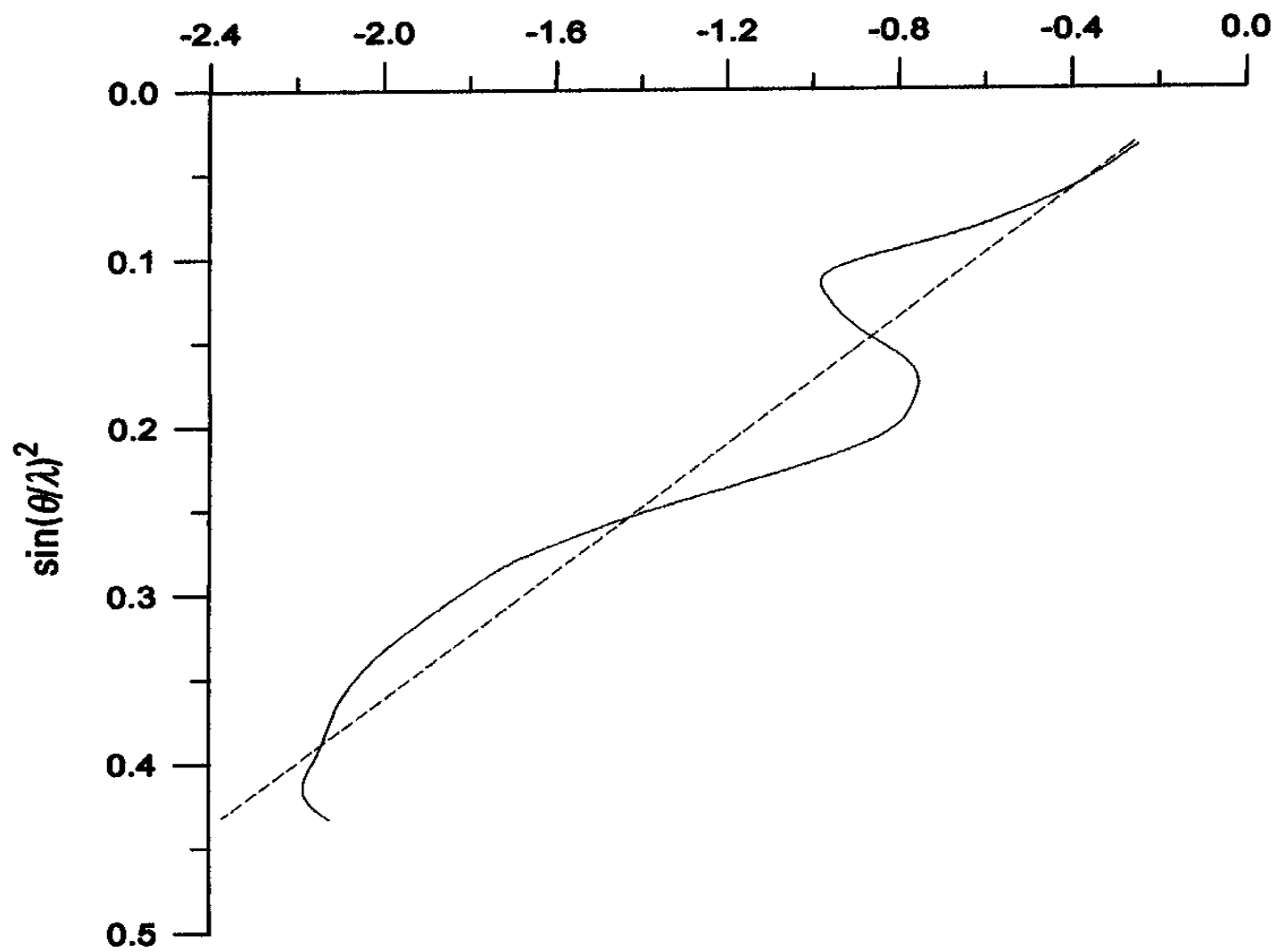
$$|F_{\mathbf{h}}|_{obs}^2 = K |F_{\mathbf{h}}|^2 = K |F_{\mathbf{h}}^0|^2 \exp(-2Bs^2)$$



$$\langle |F_{\mathbf{h}}|_{obs}^2 \rangle = K \langle |F_{\mathbf{h}}^0|^2 \rangle \exp(-2Bs^2) = K \Sigma_s^0 \exp(-2Bs^2)$$

$$\underbrace{\ln \left(\frac{\langle |F_{\mathbf{h}}|_{obs}^2 \rangle}{\Sigma_s^0} \right)}_{\mathbf{y}} = \ln K - 2Bs^2$$

\mathbf{x}

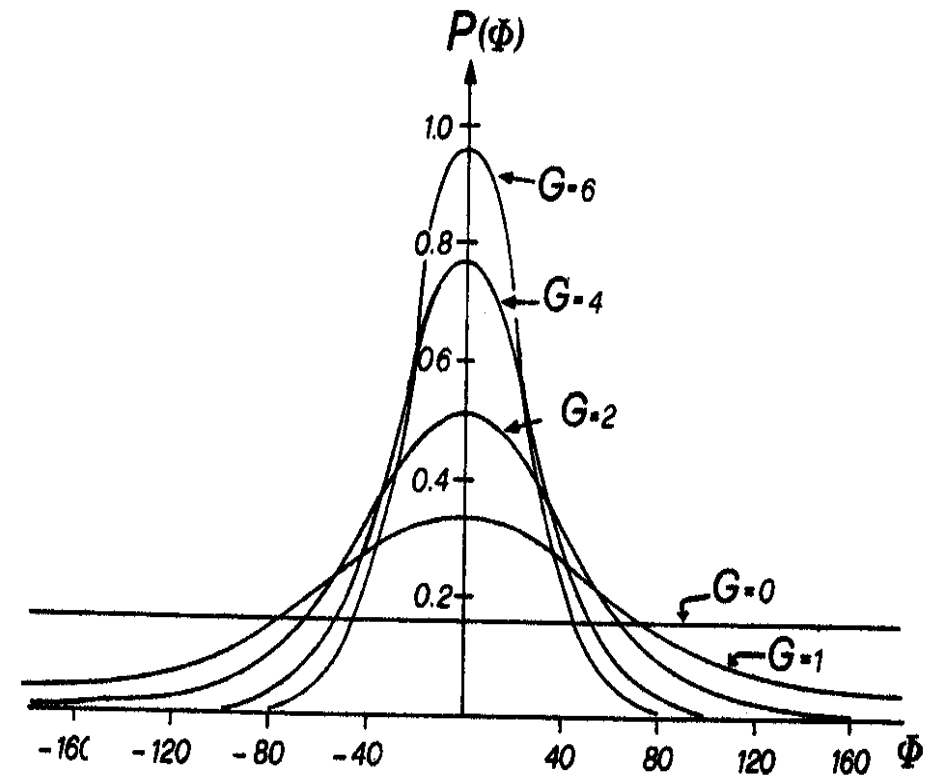


The Cochran formula

$$\Phi_{h,k} = \varphi_h + \varphi_k + \varphi_{-h-k} = \varphi_h + \varphi_k - \varphi_h$$

$$P(\Phi_{hk}) \approx [2\pi I_0]^{-1} \exp(G \cos \Phi_{hk})$$

$$\text{where } G = 2 | E_h E_k E_{h+k} | / N^{1/2}$$



Accordingly:

$$\varphi_h + \varphi_k - \varphi_{h+k} \approx 0 \quad \infty$$

$$\varphi_h - \varphi_k - \varphi_{h-k} \approx 0 \quad \infty$$

$$\varphi_h \approx \varphi_k - \varphi_{h-k} \quad \infty$$

$$G = 2 | E_h E_k E_{h+k} | / N^{1/2}$$

$$G = 2 | E_h E_k E_{h-k} | / N^{1/2}$$

$$G = 2 | E_h E_k E_{h-k} | / N^{1/2}$$

The tangent formula

A reflection can enter into several triplets. Accordingly

$$\varphi_h \approx \varphi_{k_1} + \varphi_{h-k_1} = \theta_1 \quad \text{with } P_1(\varphi_h) \propto G_1 = 2 | E_h E_{k_1} E_{h-k_1} | / N^{1/2}$$

$$\varphi_h \approx \varphi_{k_2} + \varphi_{h-k_2} = \theta_2 \quad \text{with } P_2(\varphi_h) \propto G_2 = 2 | E_h E_{k_2} E_{h-k_2} | / N^{1/2}$$

.....

$$\varphi_h \approx \varphi_{k_n} + \varphi_{h-k_n} = \theta_n \quad \text{with } P_n(\varphi_h) \propto G_n = 2 | E_h E_{k_n} E_{h-k_n} | / N^{1/2}$$

Then

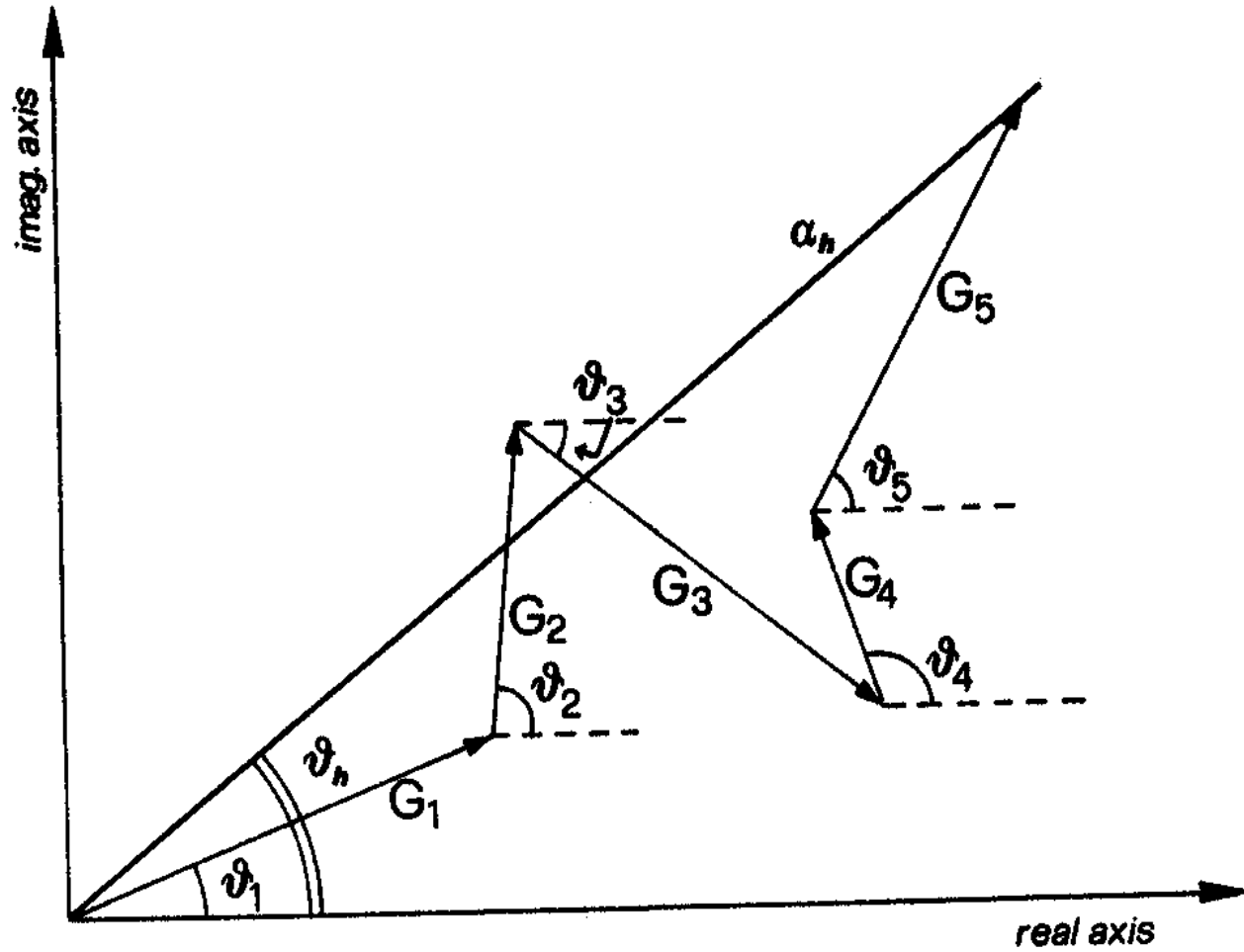
$$P(\varphi_h) \approx \prod_j P_j(\varphi_h) \approx L^{-1} \prod_j \exp [G_j \cos (\varphi_h - \theta_j)]$$

$$= L^{-1} \exp [\alpha \cos (\varphi_h - \theta_h)]$$

where

$$\tan \theta_h = \frac{\sum G_j \sin \theta_j}{\sum G_j \cos \theta_j} = \frac{T}{B}, \quad \alpha_h = \left(T^2 + B^2 \right)^{1/2}$$

A geometric interpretation of α



The random starting approach

To apply the tangent formula we need to know one or more pairs $(\varphi_k + \varphi_{h-k})$. *Where to find such an information?*

The most simple approach is the **random starting approach**. Random phases are associated to a *chosen* set of reflections. The tangent formula should drive these phases to the correct values. The procedure is cyclic (up to **convergence**).

How to recognize the correct solution?

Figures of merit can or cannot be applied

Tangent cycles

- ϕ_1 ϕ'_1 ϕ''_1 ϕ^c_1
- ϕ_2 ϕ'_2 ϕ''_2 ϕ^c_2
- ϕ_3 ϕ'_3 ϕ''_3 ϕ^c_3
-
- ϕ_n ϕ'_n ϕ''_n ϕ^c_n
-

- ***Ab initio* phasing**
- **SIR2009** is able to solve
 - small size structures* (up to 80 atoms in the a.u.);
 - medium-size structures* (up to 200);
 - large size* (no upper limit)
- It uses
- **Patterson deconvolution techniques**
- (***multiple implication transformations***)
- as well as
- **Direct methods**
- to obtain a starting set of phases. They are extended and refined *via*
- ***electron density modification techniques***
-

- **Direct methods limits for proteins:**
- 1) the large size (proteins range from 300 atoms in the asymmetric unit to several thousands). The G factor in the Cochran formula are very small.
- 2) data resolution

- To overcome the limits one is obliged to :
- -increase the **number of direct methods trials** .
The cost to pay concerns the computing time.
- - **improve and extend the the poor phases**
available by DM by exploiting some specific
features of the proteins (e.g., the solvent , etc.).

About the data resolution limit

Atomic resolution at length was considered a necessary (and not sufficient) condition for **ab initio phasing** (Sheldrick rule) , condition relaxed later on (up to 1.2 Å). If it is not satisfied:

- - the atomicity condition is violated;
- - the number of reliable triplet invariants exploitable by the tangent procedure is small.
- - Patterson and EDM procedures are less effective;
- - the small ratio
- ***number of observations/ number of parameters***
- make least squares unreliable.