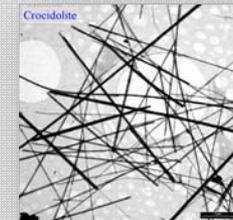
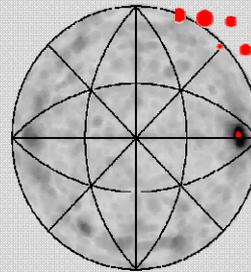
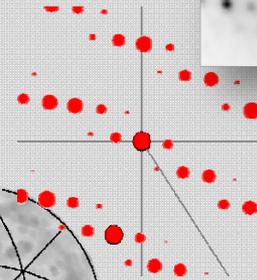
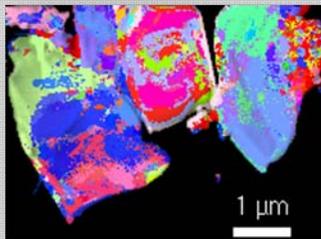
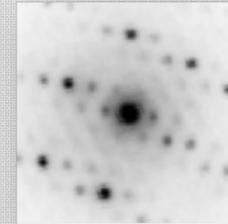
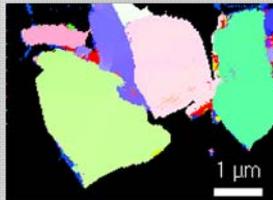
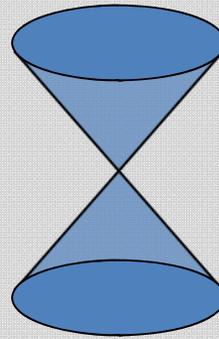
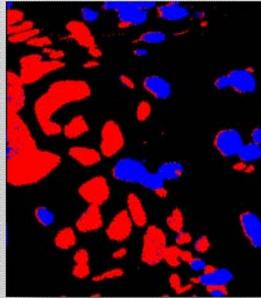
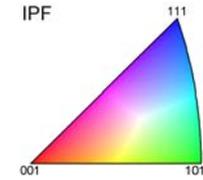
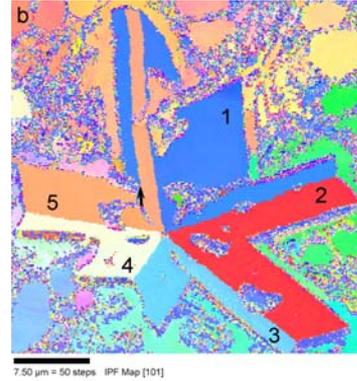
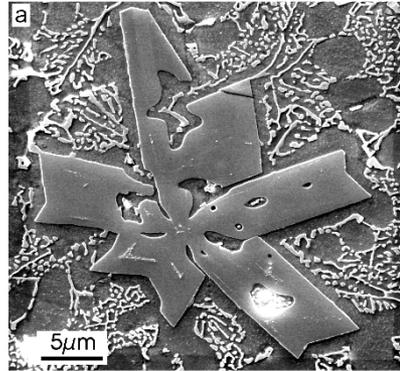


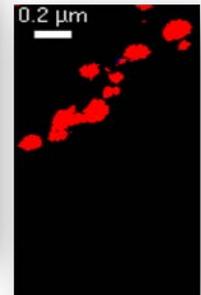
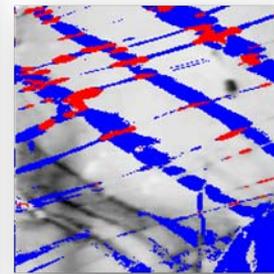
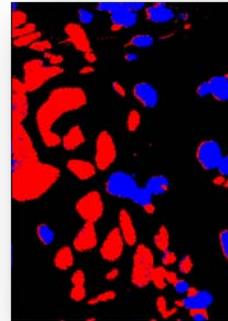
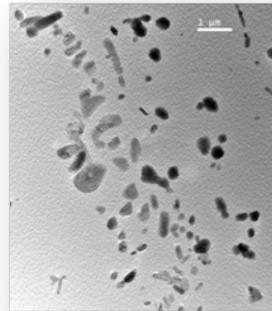
# ASTAR (EBSD-TEM like )

Automatic Crystal Orientation/Phase mapping for TEM





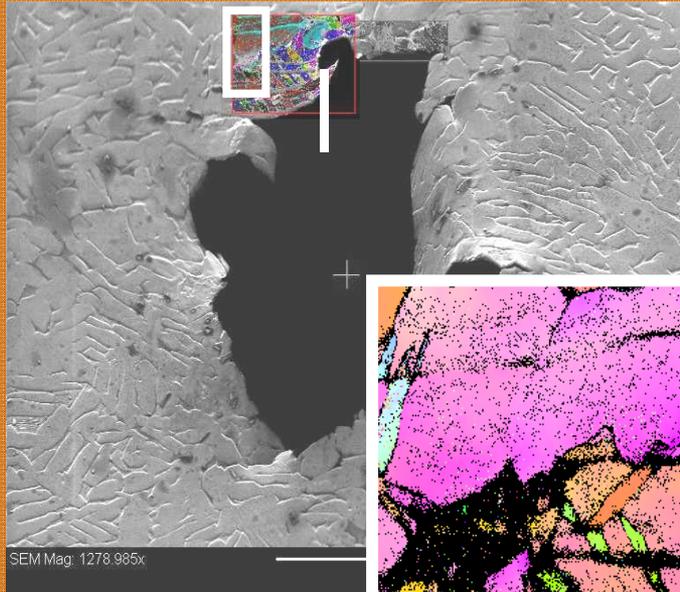
## NEW precession application “EBSD” – TEM



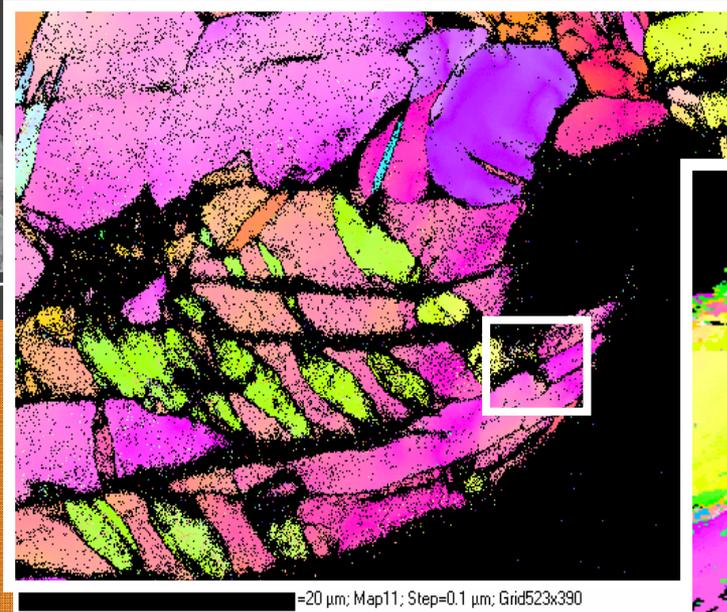
## **EBSD-TEM : beam is scanned over the sample ( eg. $10\ \mu \times 10\ \mu$ )**

- ➔ spot electron diffraction patterns are collected ( NOT sensitive to stress/strain or surface sample preparation like in EBSD-SEM )**
- ➔ Beam scanning performed by “spinning star” unit / no STEM need**
- ➔ Thousands of experimental spot ED patterns are acquired by a very fast optical CCD camera attached to TEM screen ( 180 patterns/sec )**
- ➔ Slow scan CCD can also be used ( but slow : 20-30 patterns/sec )**
- ➔ Thousands of theoretical ED patterns are generated ( templates ) from .cif files or commercial databases for all known phases in a sample**
- ➔ Template matching is used ( by cross-correlation of all experimental ED patterns with all templates ) to generate most probable orientation of every scanned position in the sample.**

# Comparison SEM-(EBSD) vs TEM spatial resolution

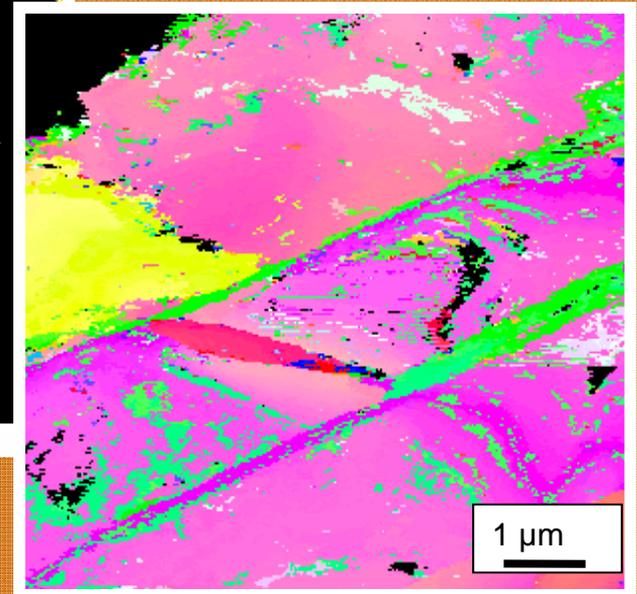


SEM orientation map  
deformed Ta<sub>6</sub>V alloy



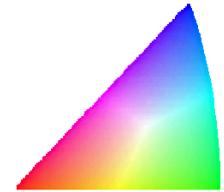
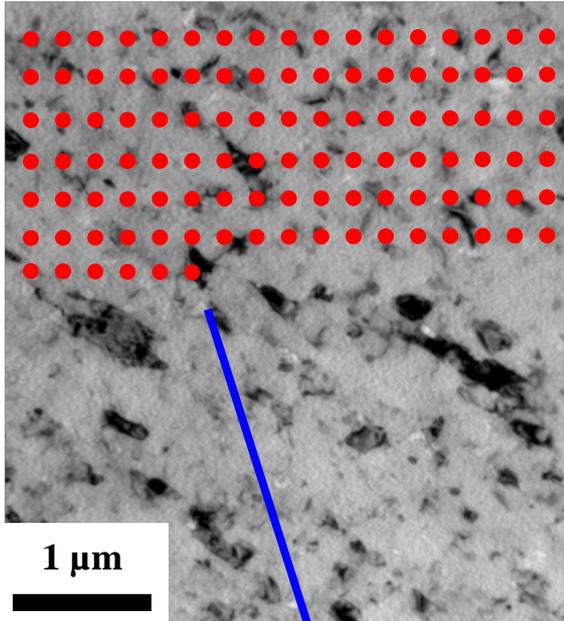
EBSD map (100 nm stepsize)

TEM orientation map  
(25 nm stepsize)

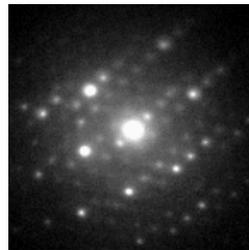


Electron Backscattering Diffraction (EBSD) orientation maps in SEM have usually poor resolution in comparison with TEM maps showing detailed nanostructure

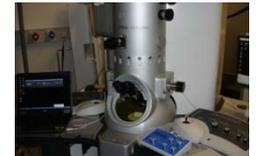
# ASTAR : diffraction pattern adquisition



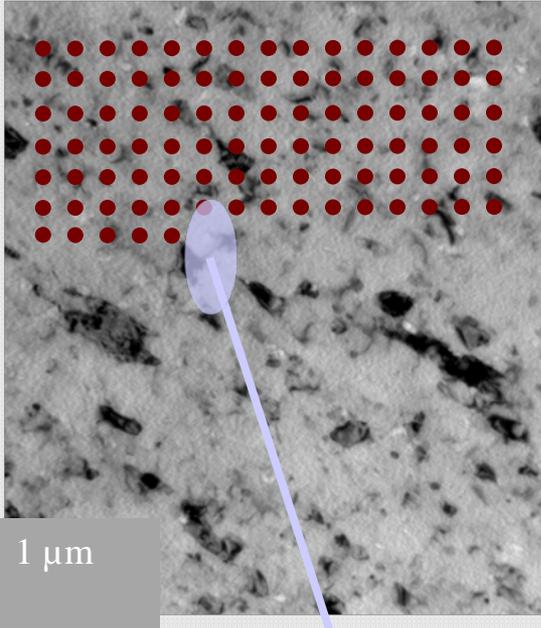
Example :Severely  
deformed  
7075 Aluminium Alloy



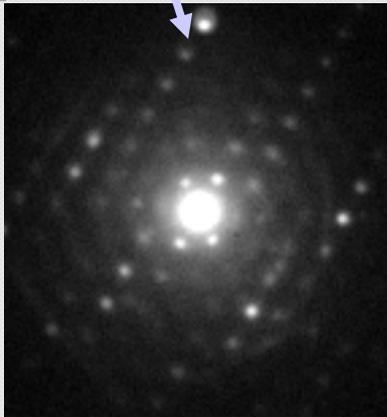
Any TEM -FEG/LaB6  
may work with ASTAR



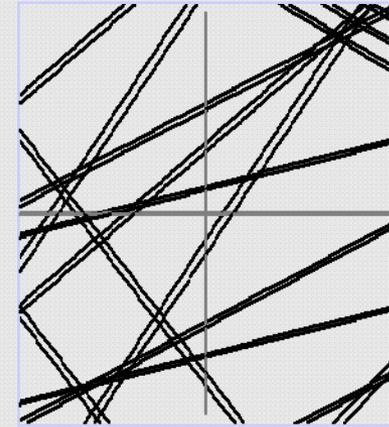
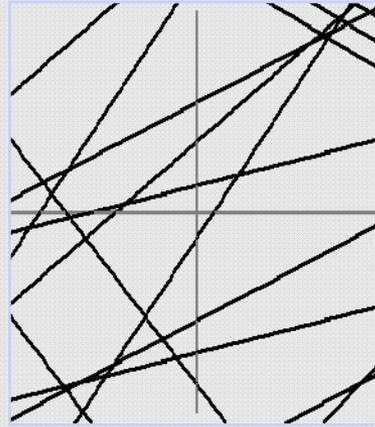
# EBSD-TEM : Automated Crystal Orientation Mapping



Severely deformed  
7075 Al. Alloy

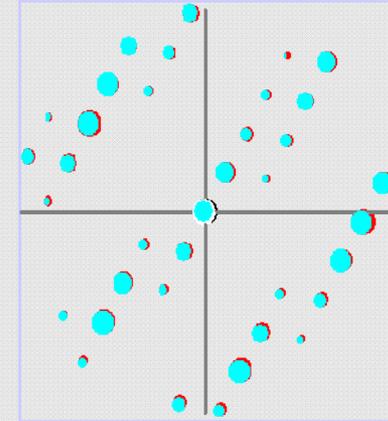
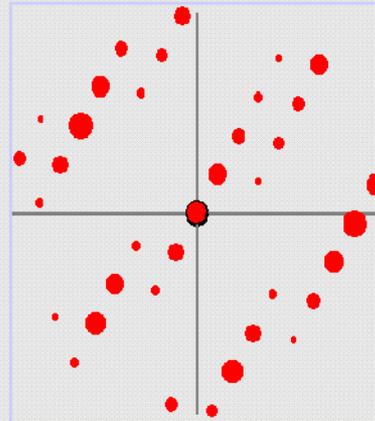


Kikuchi pattern



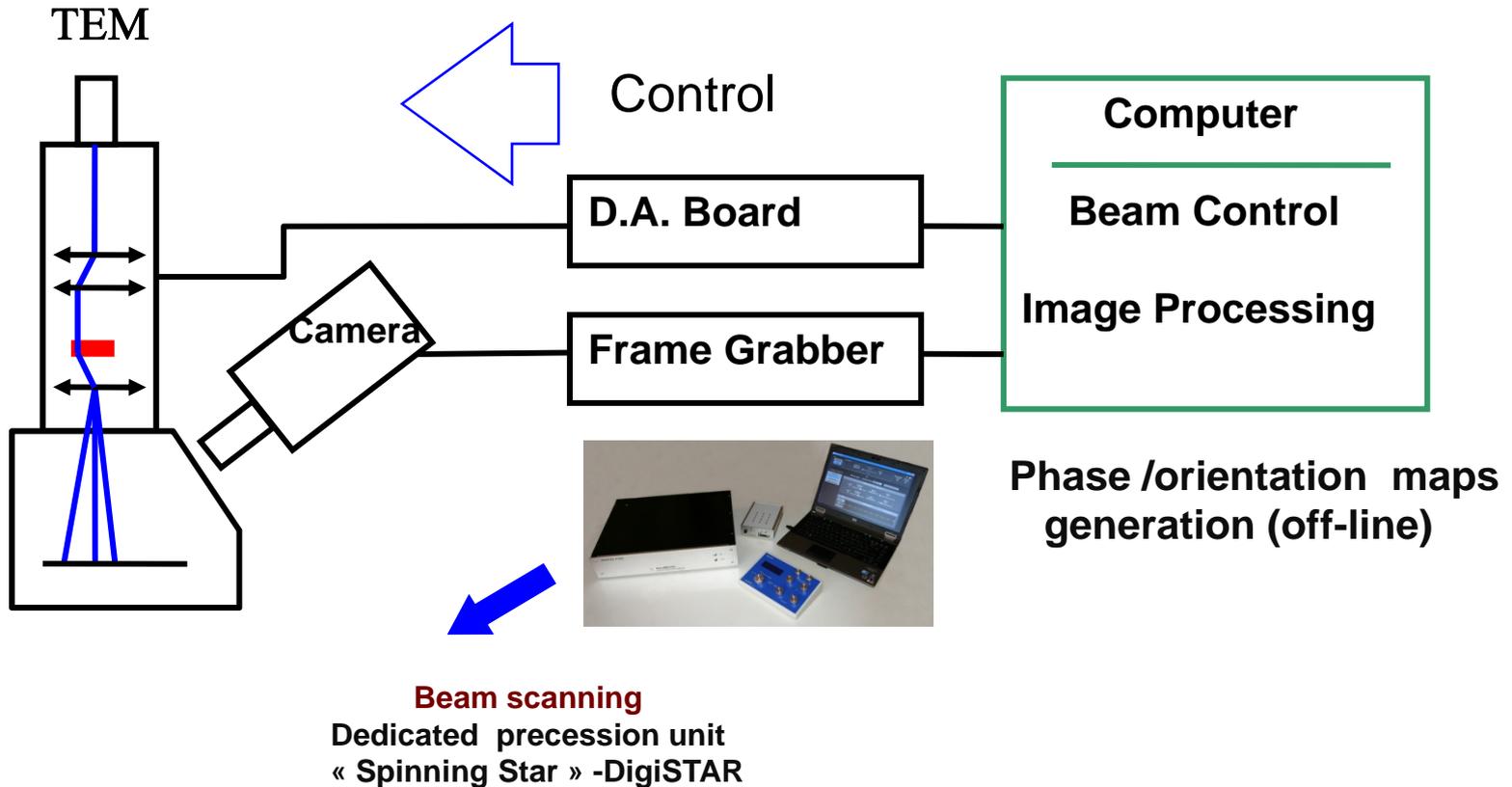
Orientation  $\Omega$

Orientation  $\Omega + \Omega' (= \Omega + 0.1^\circ)$

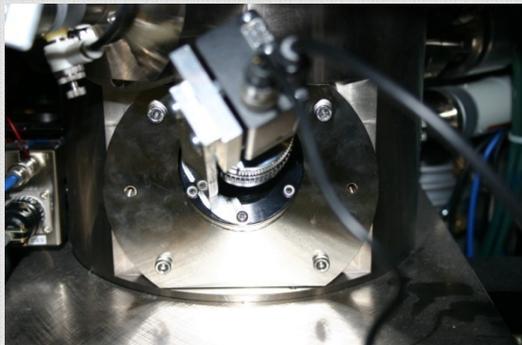
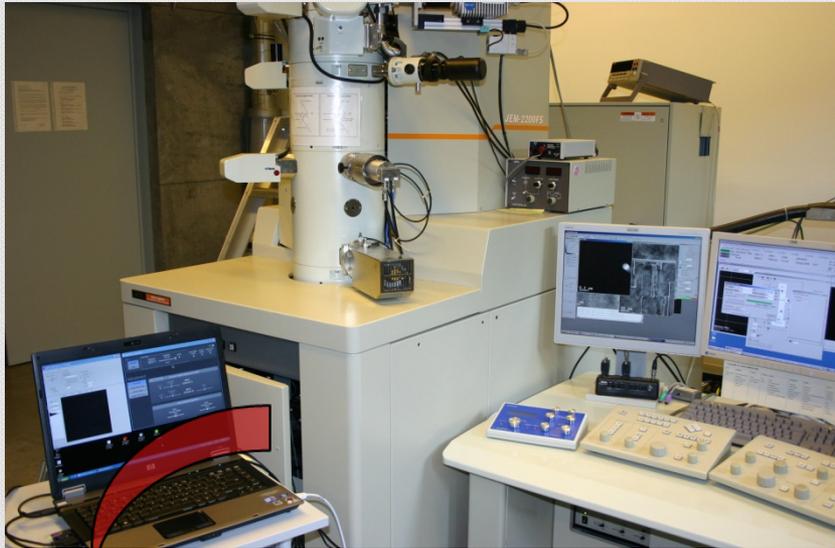


Bragg Spot pattern

# ASTAR ( EBSD-TEM Procedure )



**ASTAR : Automatic Crystal Orientation  
and  
phase mapping hardware/software package for TEM**



**Dedicated fast CCD camera (> 100 patterns/sec) attached to the TEM screen**

# DiffGen : Template generator

The screenshot displays two windows from the DiffGen software. The left window, titled 'Diffraction pattern', shows a 2D diffraction pattern with red spots on a grid. The right window, titled 'Cell structure editor', shows the crystal structure parameters for 'Aluminium oxide (2.667/4) - gamma'.

**Diffraction pattern window:**

- Structure Orientation Pattern Create Bank
- Phi1: 319.79, PhiM: 57.89, Phi2: 62.92
- Orientation:  4 Miller's indices
- Approximated hkl: 6 3 4 Err. 1.37°
- Display control: Spot size 5, Count 5137, Zoom level 100, Rotat. step 1
- ✓ Show Kikuchi lines 6,  Show spots indices,  Hide 0,0,0
- Symmetry: Laue Class 4/mmm, Current symmetry 0
- ✓ FPI / FP
- Diffraction settings: Wave length (Å) 0.02, Max angle (°) 3, Excitation error 1
- Intensity control:  Double diffraction, Intensity scale 9, Minimum intensity 0.1

**Cell structure editor window:**

- Import Structure Export Structure
- Structure name: '99836-ICSD 'Aluminium oxide (2.667/4) - gamma'
- Parameters: a 5.652, b 5.652, c 7.871, alpha (°) 90.00, beta (°) 90.00, gamma (°) 90.00, Space group 141 141/m m d, Laue class 4/mmm, Atoms count: Base 4, Total 28
- Enter base atoms, compute equivalents (selected) / Enter all atoms manually
- Table of atoms:

	Elem.	Occ	F	x	y	z	Int
1	Al1	0.78	13	0	0.75	0.125	0.780
2	O1	1	8	0	0.0076	0.2516	0.620
3	Al3	0.58	13	0	0	0.5	0.580
4	Al2	0.36	13	0	0	0	0.360
5	O1	1	8	0.2424	0.75	0.5016	0.620
6							0.20
7							0.20
8							0.20
9							0.20
10							0.20
11	O1	1	8	0.2576	0.75	0.9984	0.620
12	O1	1	8	0.5	0.0076	0.2484	0.620
13	O1	1	8	0.2424	0.25	0.4984	0.620
14	O1	1	8	0.7424	0.75	0.9984	0.620
15	O1	1	8	0.7576	0.25	0.4984	0.620
16	O1	1	8	0.5	0.5076	0.7516	0.620

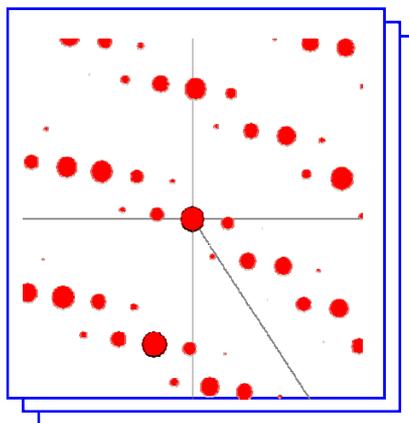
**Equation:**

$$F_{hkl} = \sum_i f_i e^{2\pi i (hx_i + ky_i + lz_i)}$$

Features: Any crystallographic structure  
 Laue class adapted to the space group  
 Structure generator (space group, structure factor equ.)

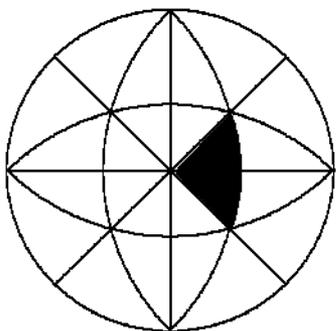
# ASTAR : crystallographic orientation identification

Pre-calculated templates

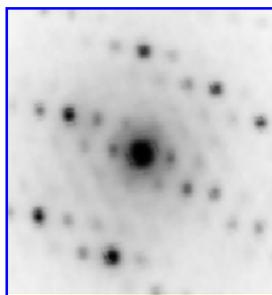


$$Q(i) \sim \sum_{j=1}^m P(x_j, y_j) T_i(x_j, y_j)$$

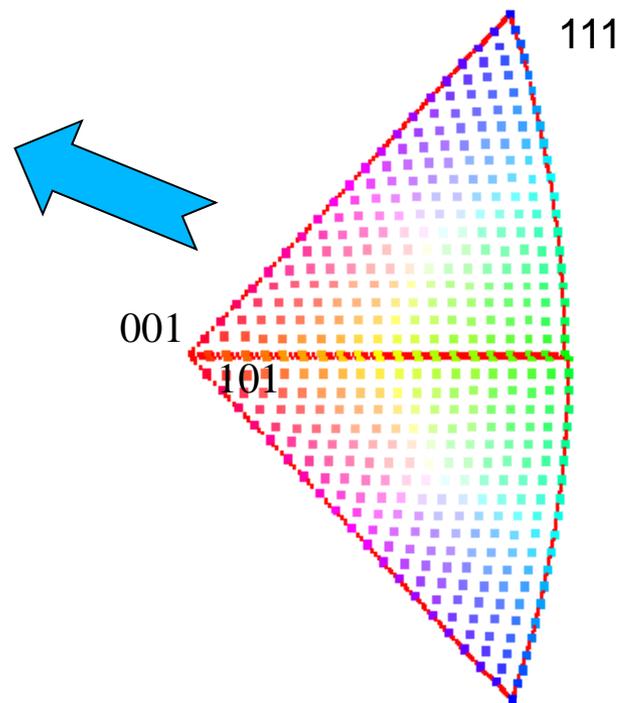
Correlation index



Acquired pattern



Template generation of all possible simulated orientations (every 1°) within stereographic triangle for given crystal lattice(s) and symmetry



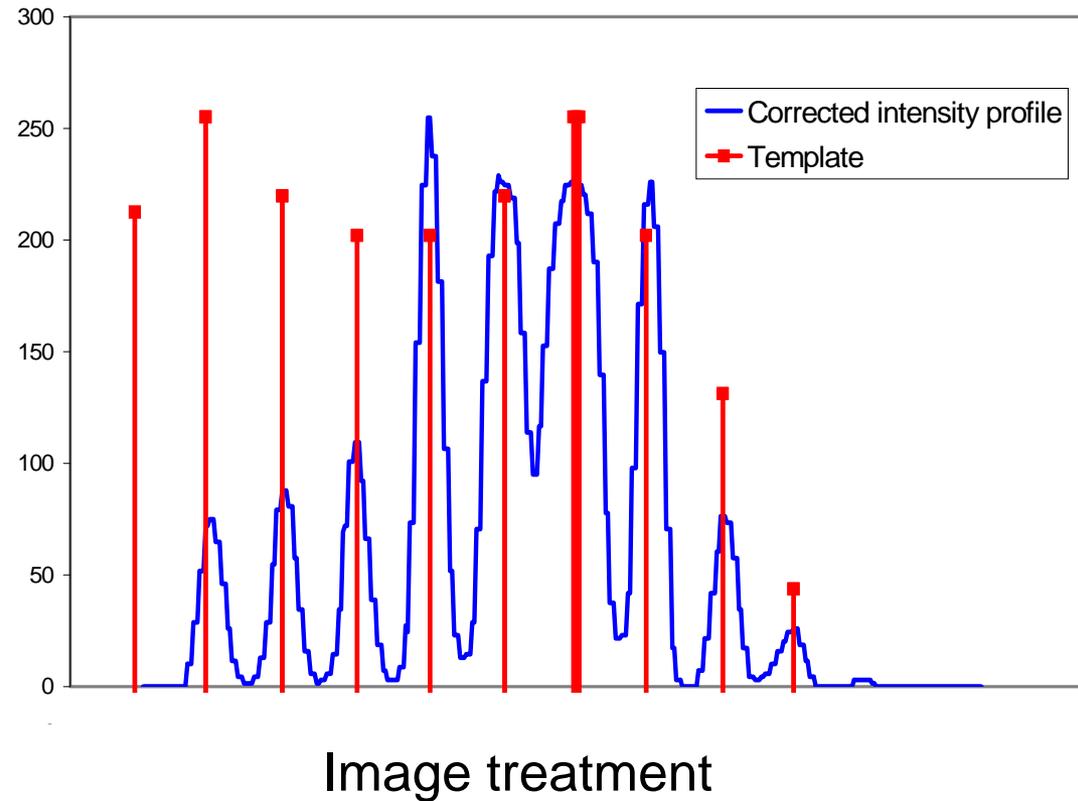
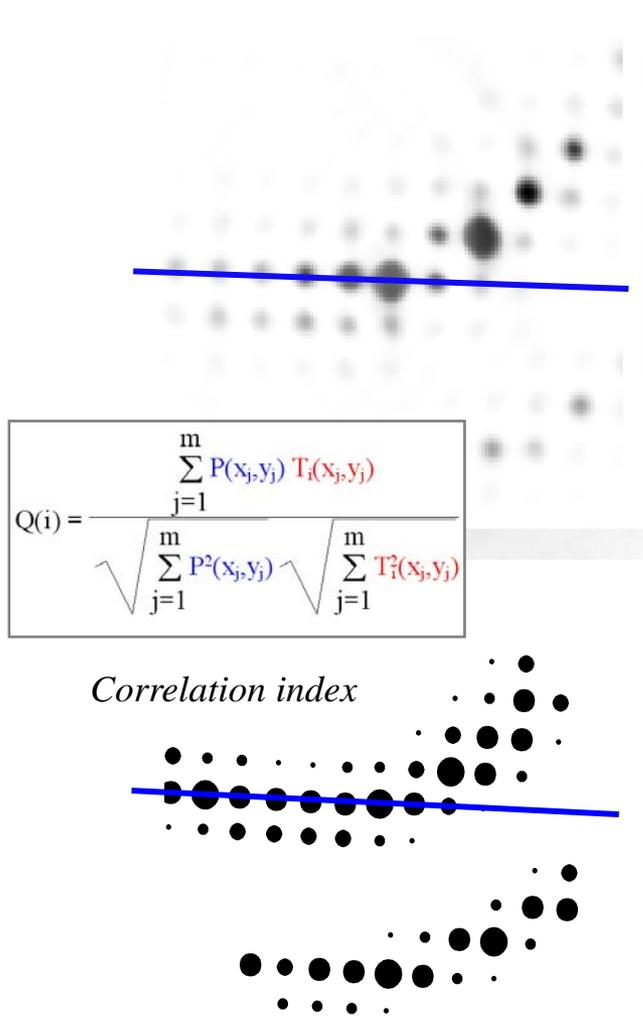
Stereographic projection

1-11

Degree of matching between experimental patterns and simulated templates is given by a correlation index ; highest value corresponds to the adequate orientation/phase

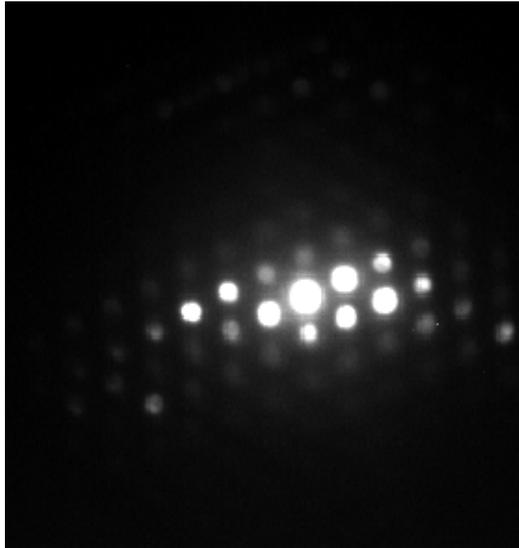
(example ,cubic)2000~ **simulated patterns**

# ASTAR : pattern matching by image cross- correlation

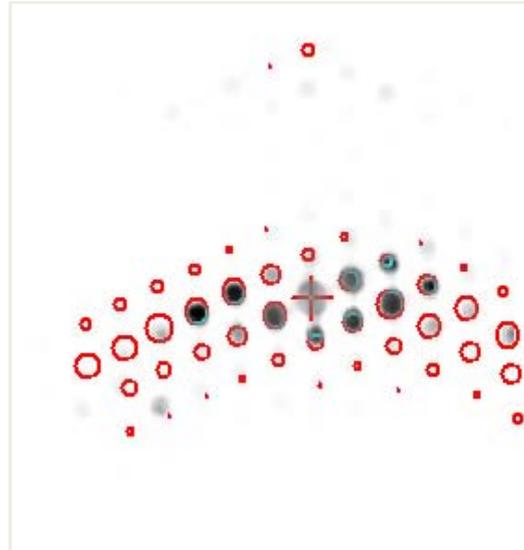


**Cross-correlation comparison of all acquired ED patterns with all simulated templates to deduce correct pattern index;** degree of matching between experimental patterns and simulated templates is given by a correlation index where highest value corresponds to the adequate orientation/phase.

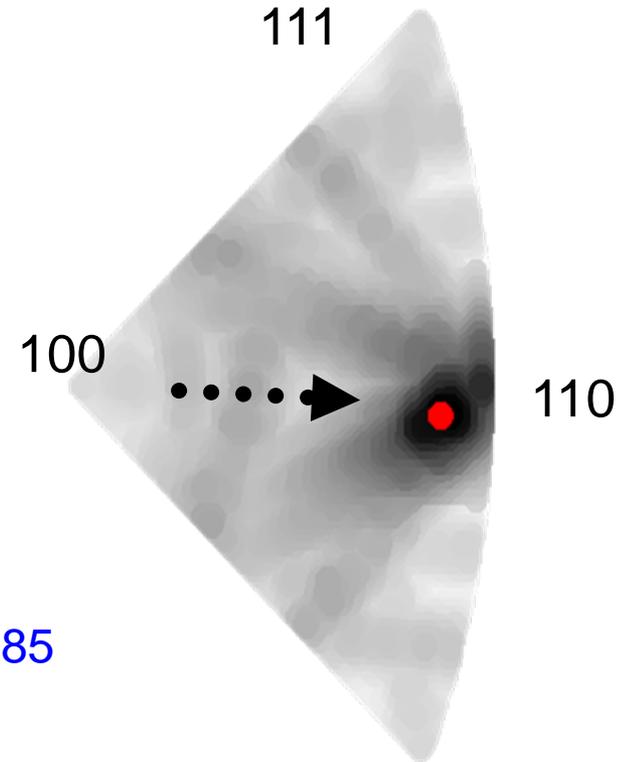
# ASTAR identification example : nanocrystalline Cu



Diffraction pattern  
( nanocrystalline cubic copper)



correlation index = 585



**Correlation index map**

For a given ED pattern, the correlation index map is calculated for all possible template orientations and plotted on a map that represents a portion of the stereographic projection (reduced to a double standard triangle). That resulting map reveals the most probable orientation for every experimental spot ED pattern ( in this case ED pattern is found to be close to 110 ZA orientation )

# ASTAR : ultra-fast TEM orientation map

Sample : severely deformed copper



Orientation map

250 x 200 pixel data acquisition

**5 min !!**

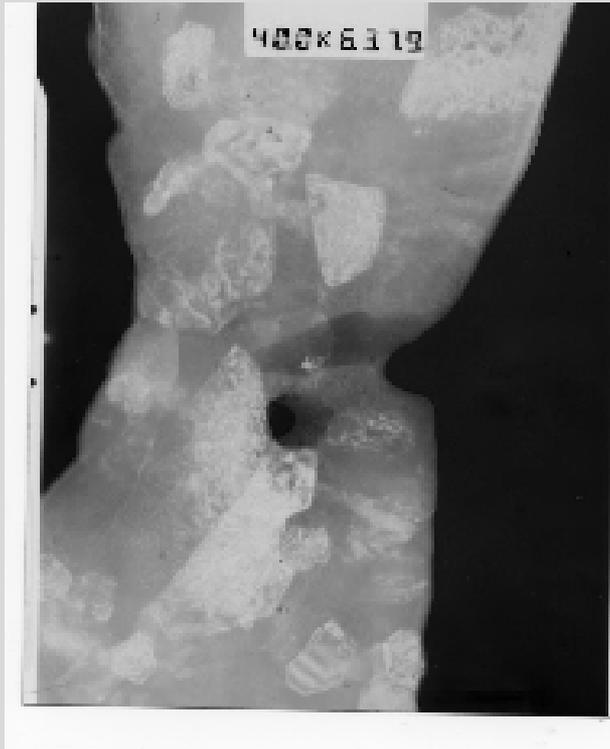
Typical software data analysis  
time ( for cubic )

**5-15 min**

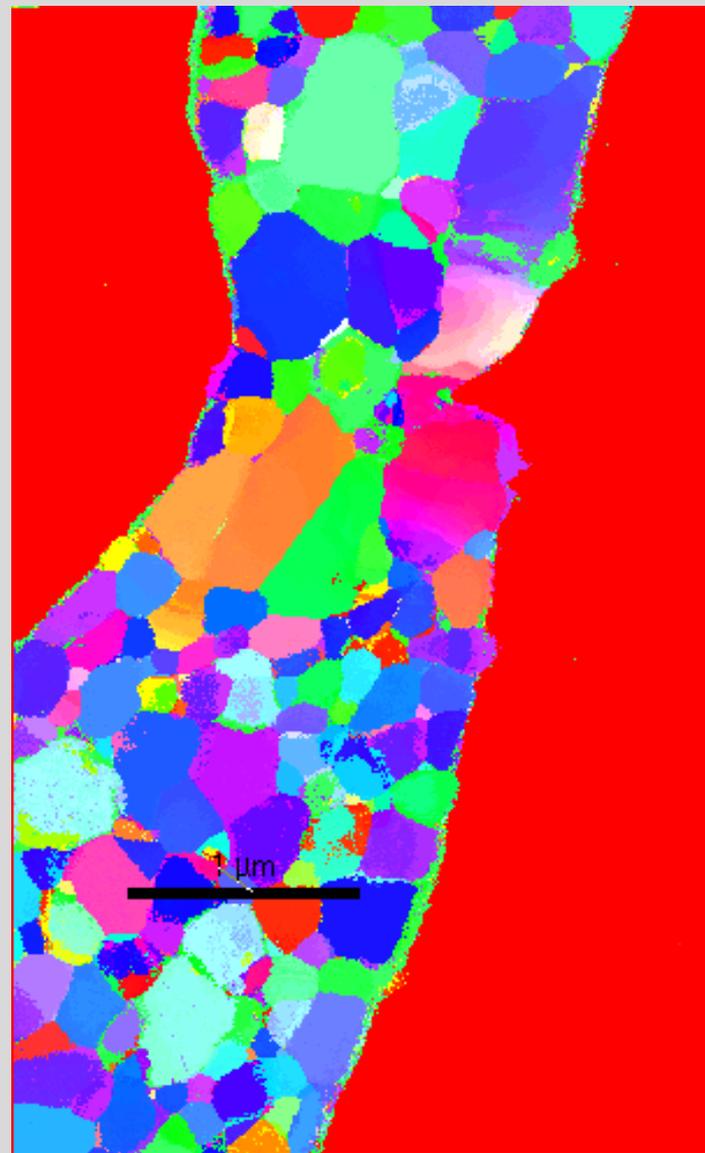
( hexagonal , tetragonal )

x 3- 4 more time

NBD step 20 nm



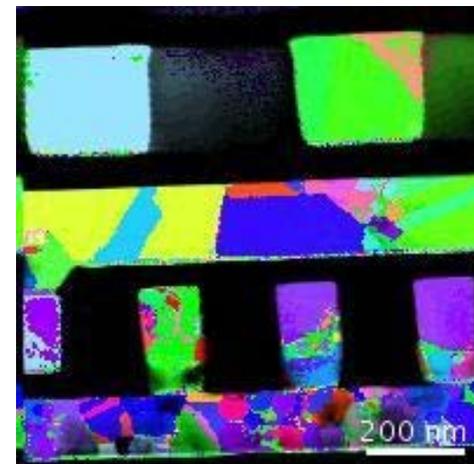
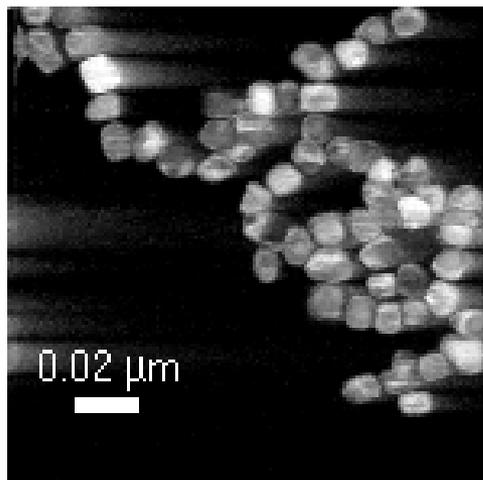
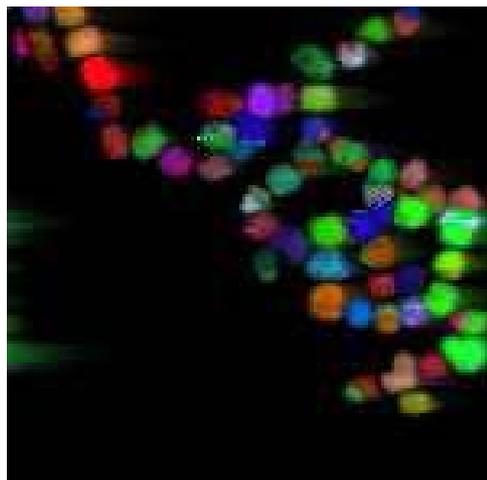
0.5 $\mu$ m



- Comparison of TEM image, and ASTAR results **15 nm resolution**
- CM20 UTwin LaB6 courtesy Prof . S.Godet ULB Brussels

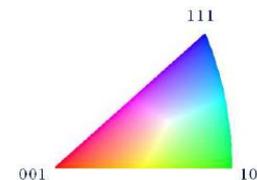
# Pt nanoparticles , Cu lines in semiconductors ( FEG –TEM )

Orientation (EBSD-like maps) **at 1 nm resolution !**



Courtesy Jeol Japan- Jeol 2100F

## Pt nanoparticles

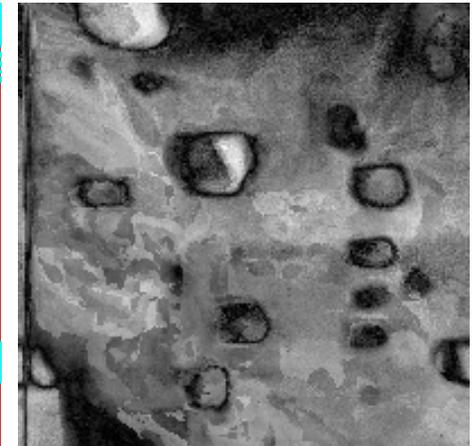
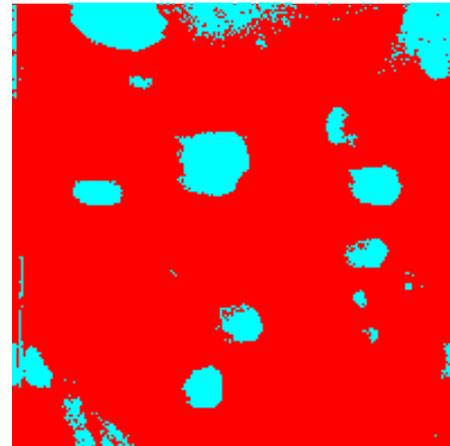
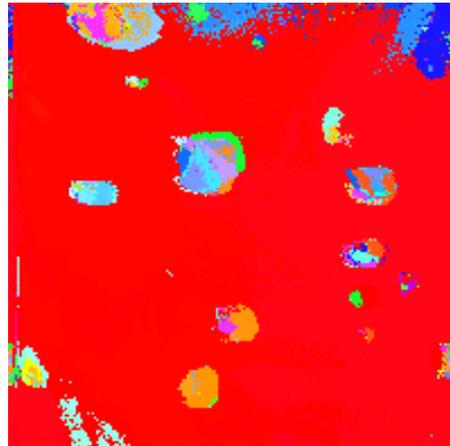
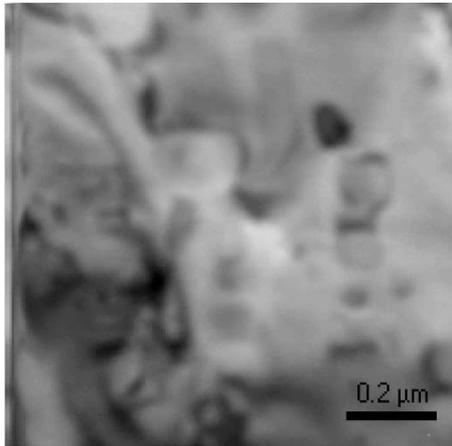


**ASTAR** phase/orientation TEM device with Jeol 2010 FEG , Jeol 2200 FS , **1 nm spot size NBD mode**, 150x150 pixel, **step size 1 nm** , 15 min acquis. time

Courtesy Prof. P.Ferreira , Ganesh Univ Texas at Austin  
Dr. Holm Kirmse Humboldt Univ Berlin



# (Mn ,Ga) As clusters in GaAs matrix

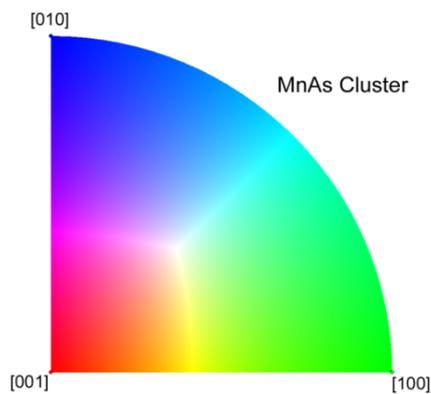


Virtual bright field map

Orientation map

Phase map

Reliability map

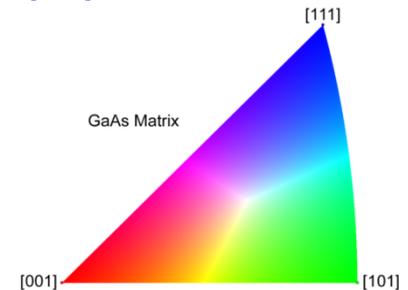


**GaAs cubic F-43m (216) a= 0.56533 nm**

**MnAs  $\beta$  phase orthorhombic P2<sub>1</sub>/n2<sub>1</sub>/m2<sub>1</sub>/a (62)**

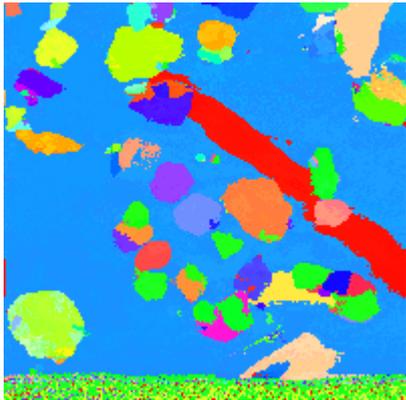
a= 0.5704 nm, b= 0.3655 nm, c= 0.6365 nm

*MnAs  $\alpha$  phase hexagonal P6<sub>3</sub>/mmc*

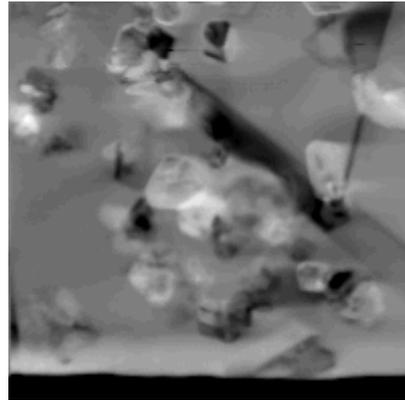


Si (matrix) and SiC  $\beta$  3C ( $a= 0.436$  nm)

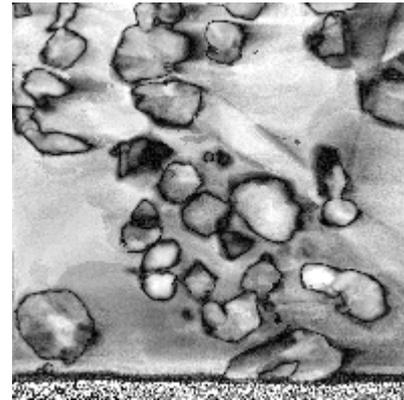
## Which orientation relation between matrix – precipitates?



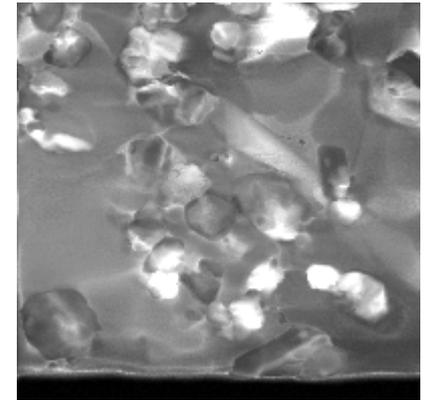
Orientation map



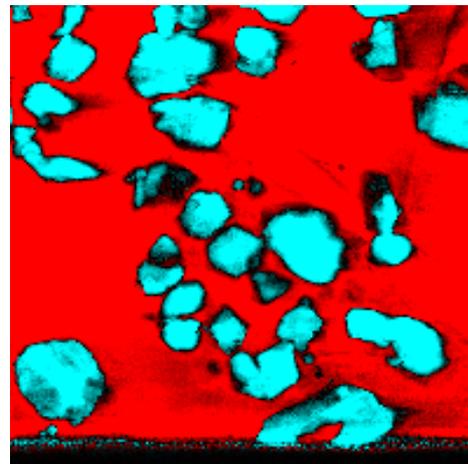
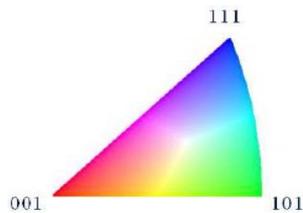
VBF



Reliability



Index



Phase map: Si red, SiC blue

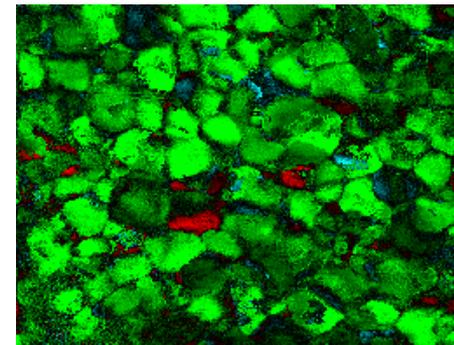
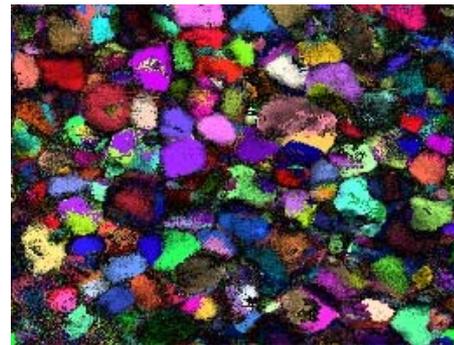
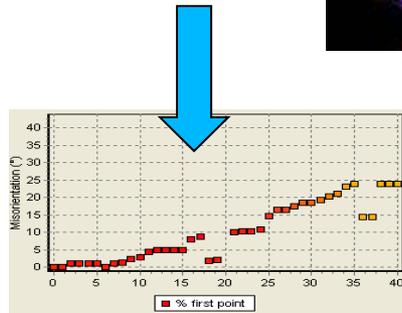
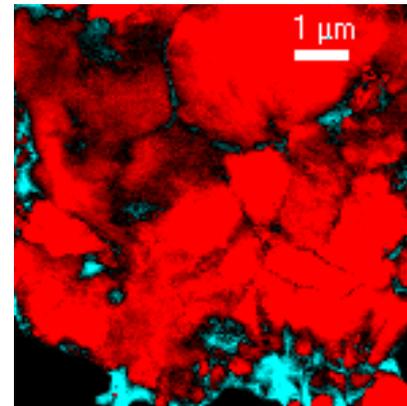
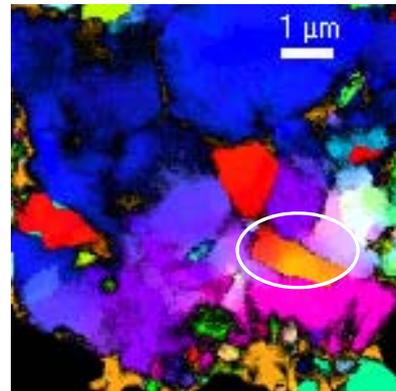
Si cubic  $a= 0.5428$  nm

**Grain size 100-500 nm**

Jeol 3010, 25 nm spot size



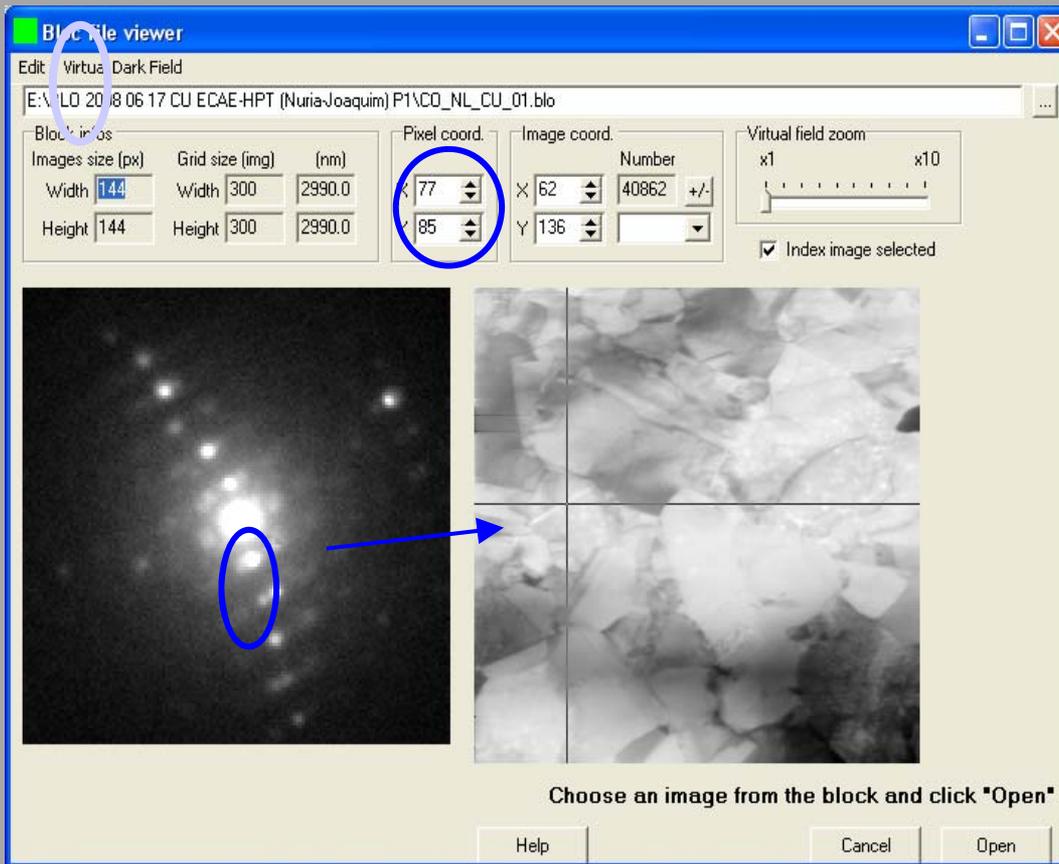
Orientation (a) and phase map of TiNb alloy revealing **cubic phase** (in red  $a = 0.332 \text{ nm}$   $Im-3m$ ) and **orthorhombic phase** (in blue  $a = 0.3215 \text{ nm}$ ,  $b = 0.485 \text{ nm}$ ,  $c = 0.462 \text{ nm}$   $Cmcm$ )



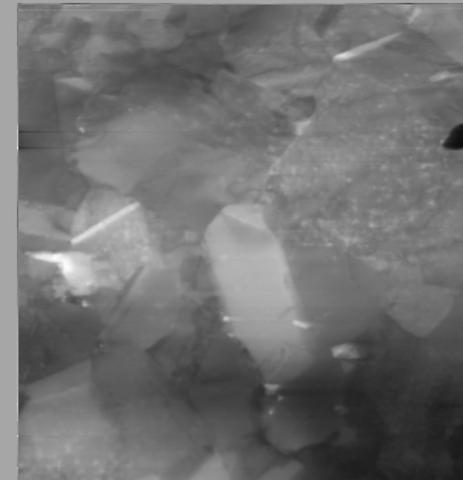
Orientation (c) and phase map (d) of AlCu alloy (in green AlCu monoclinic phase  $C2/m$   $a = 1.206 \text{ nm}$ ,  $b = 0.410 \text{ nm}$ ,  $c = 0.691 \text{ nm}$ ,  $\beta = 54,04^\circ$ , in red  $Al_3Cu_4$  orthorhombic phase  $Fmm2$   $a = 0.812 \text{ nm}$ ,  $b = 1.419 \text{ nm}$ ,  $c = 0.999 \text{ nm}$ )  
courtesy Prof.V.Demange V.Dorcet, Univ Rennes France

# INDEX : pattern identification software

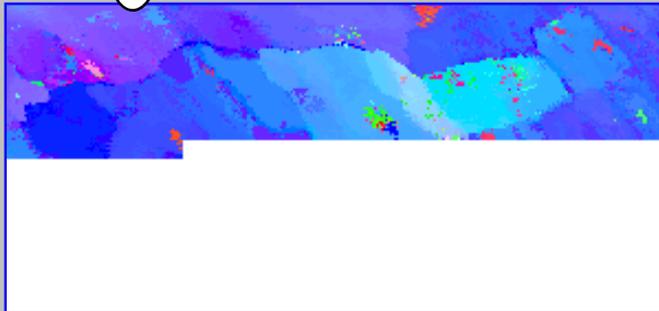
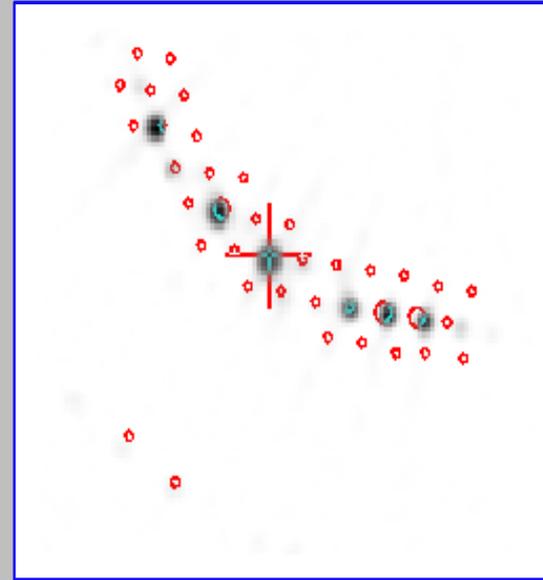
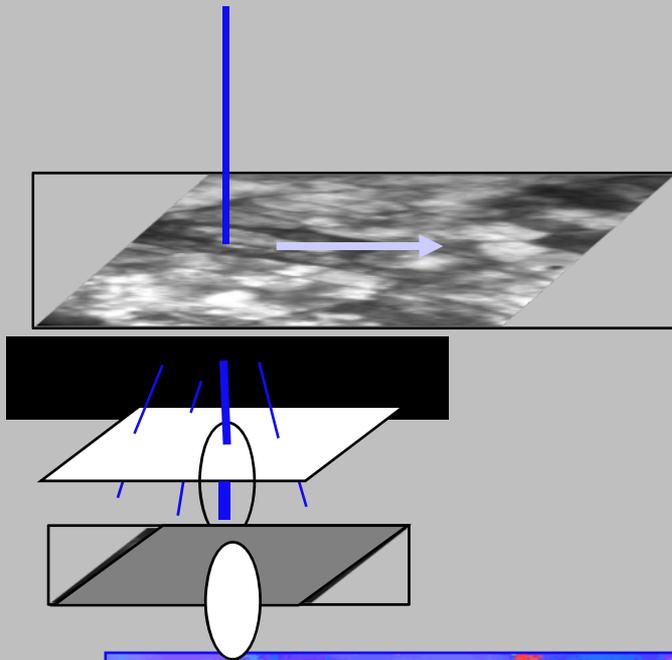
## Diffraction Pattern Block viewer



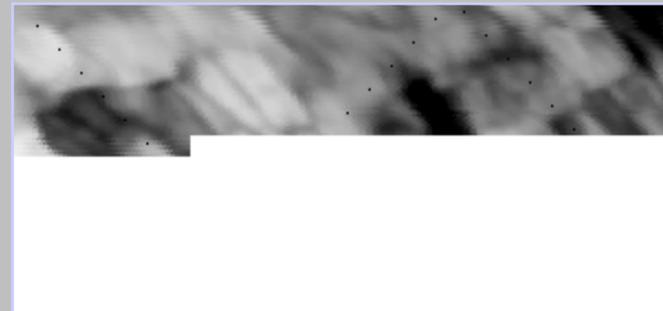
Virtual dark field image



# TEMdpa : Virtual Bright Field on-line construction

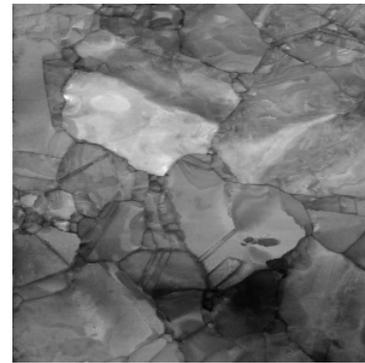
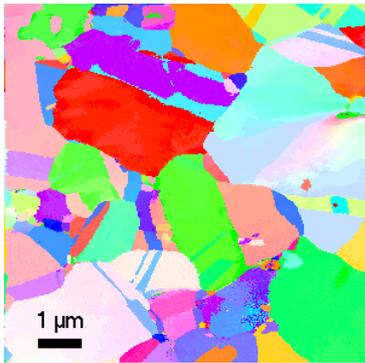


Orientation map



Bright field

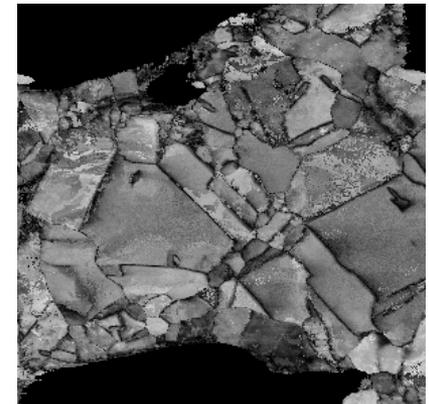
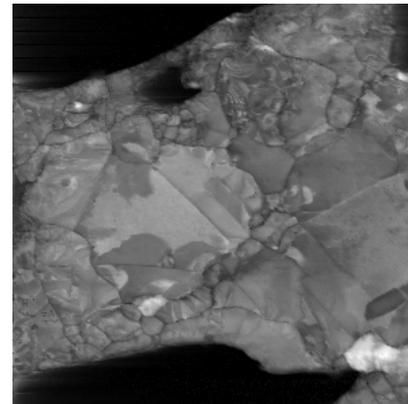
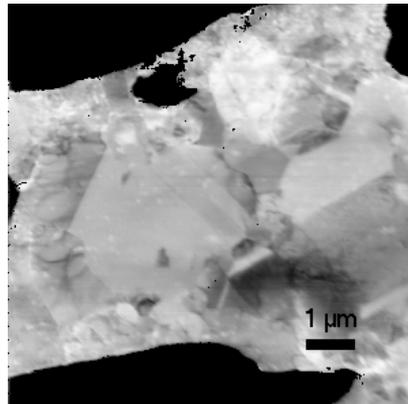




VBF

index

reliability



VBF

index

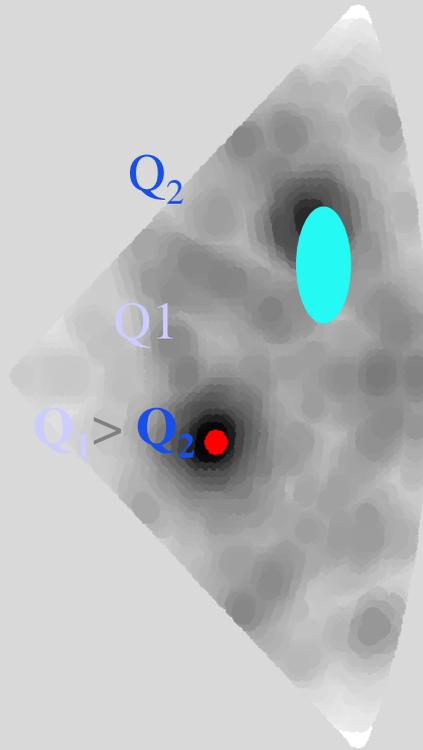
reliability

**ASTAR (EBSD -TEM) orientation maps : Nanotwins in Cu**

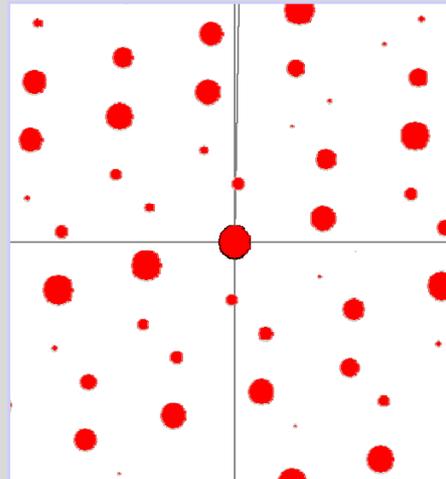
**CBD mode Jeol 3010 microscope**

# ASTAR : Reliability

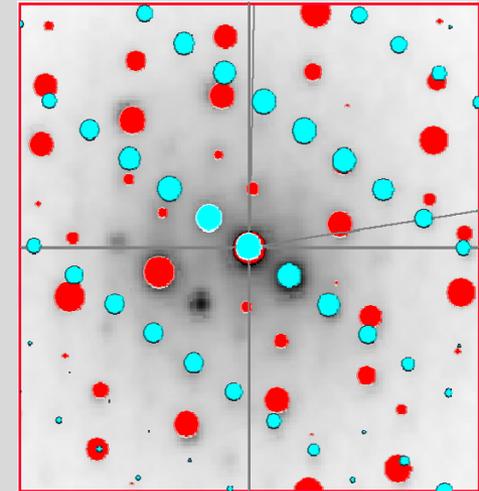
Stereographic projection



Templates for copper



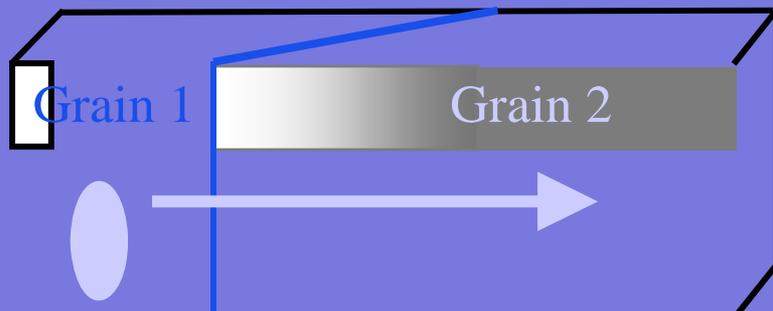
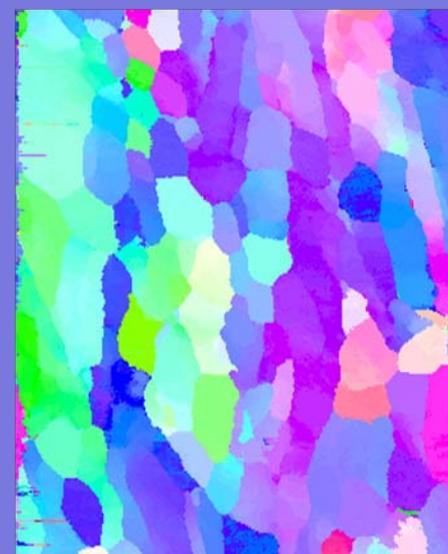
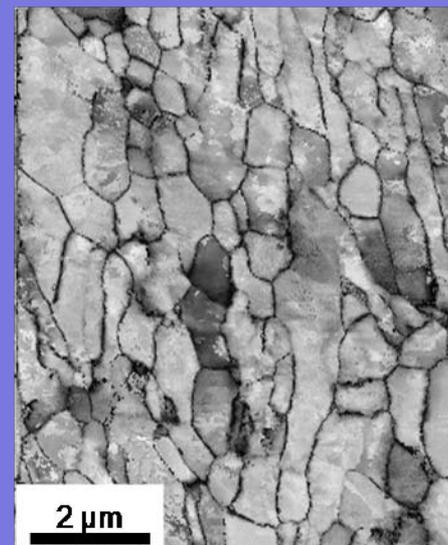
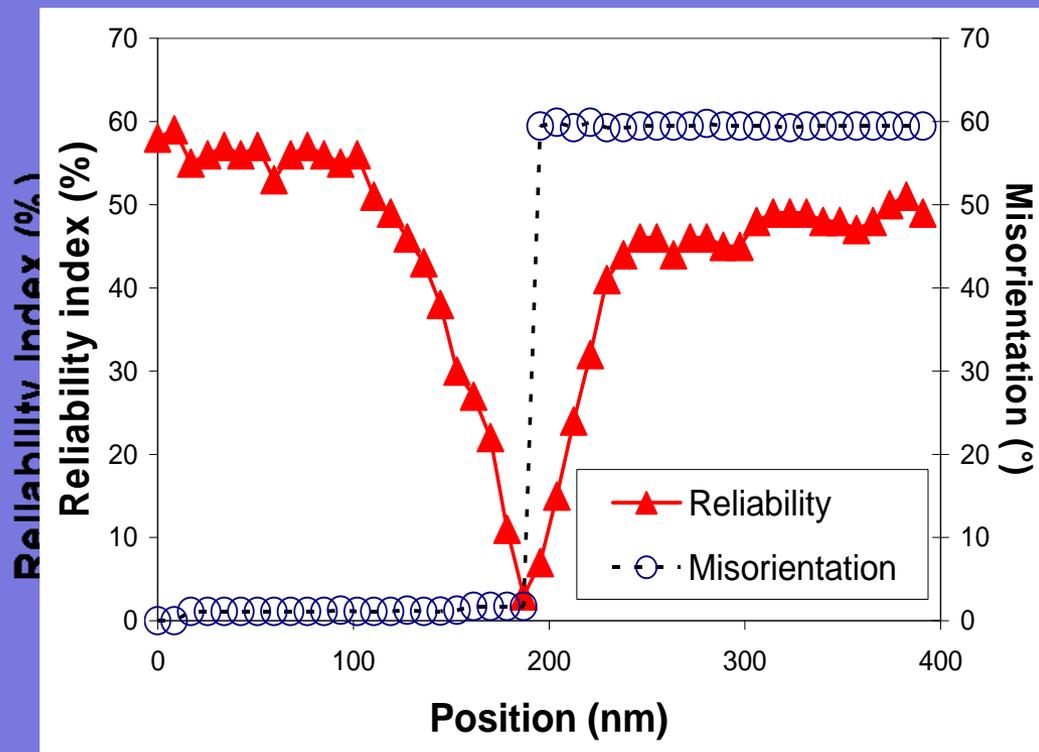
Superimposed diffraction patterns at a grain boundary



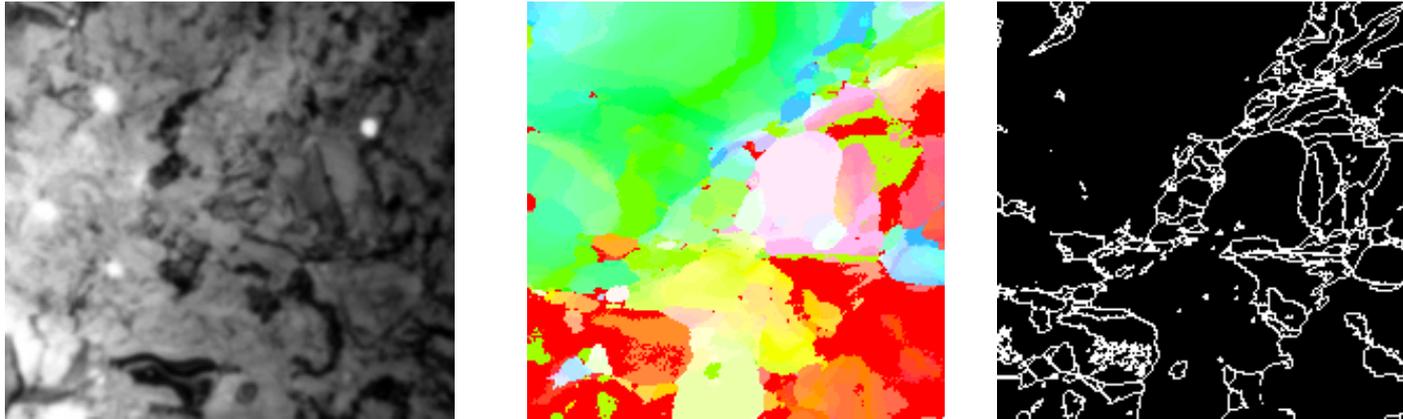
$$R = 100 (1 - Q_2/Q_1)$$

**Reliability**

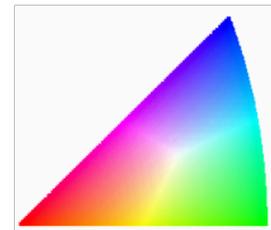
# Deconvolution of superimposed Diffraction patterns



# Imaging Grain Boundaries



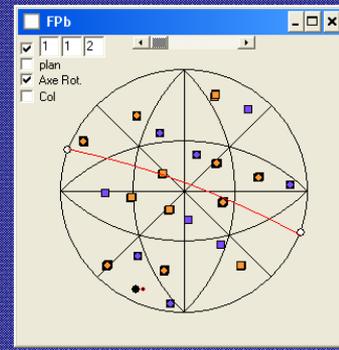
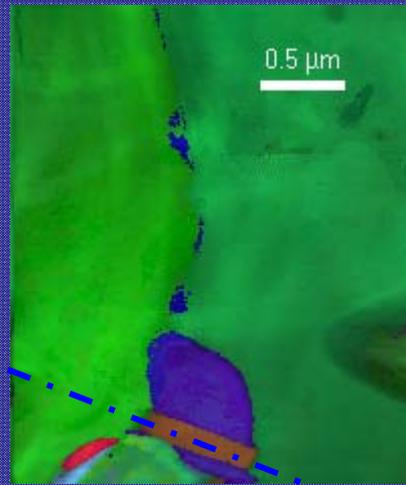
Deformed Zr alloy(Zirkaloy 4 rolled down 78%) bright field TEM image (left) showing poor grain boundary contrast, area  $5 \times 5 \mu\text{m}$  (center) same area ASTAR orientation map (right) **highlighted grain boundaries with angle  $> 15^\circ$**



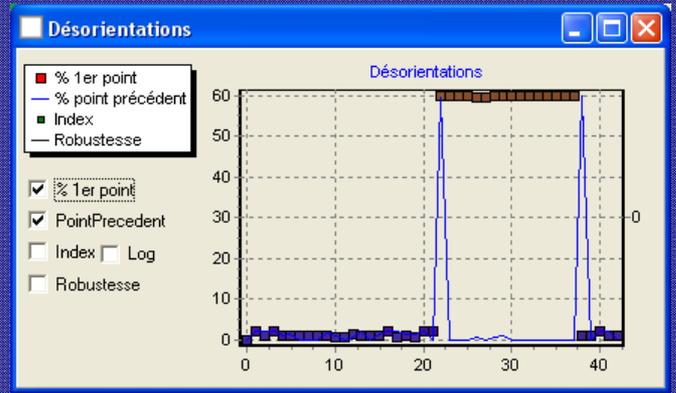
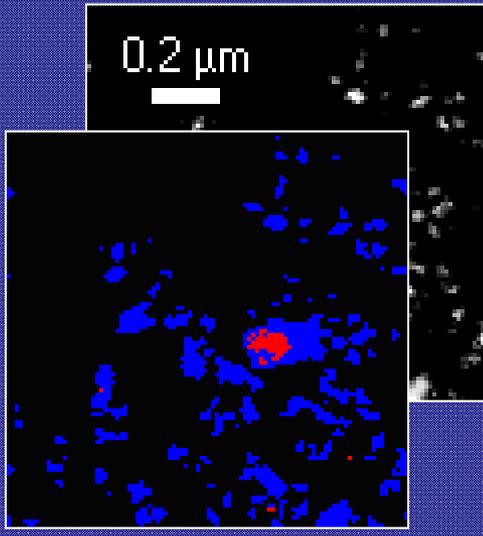
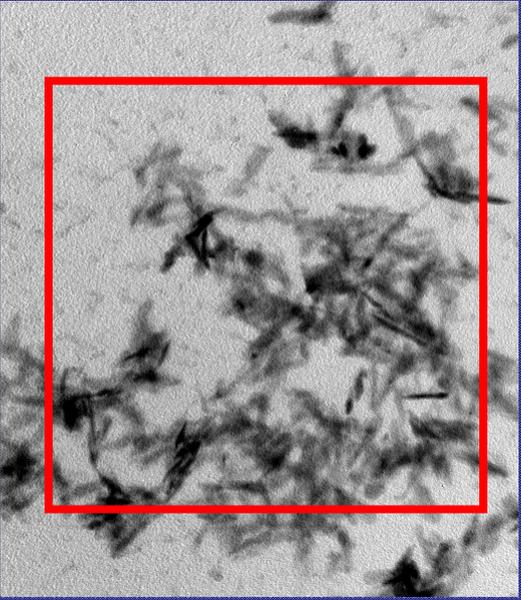
Courtesy Nippon Steel

Additional information obtained during this experiment :

### Study of Twins



### Study of Precipitates



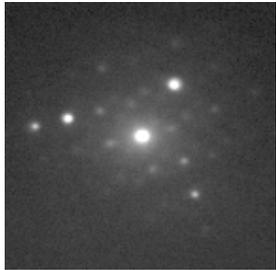
Identification of the various precipitates :

- Cementite (blue)
- TiN (red)

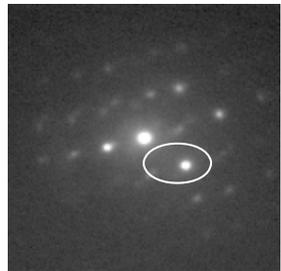
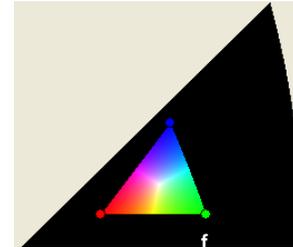
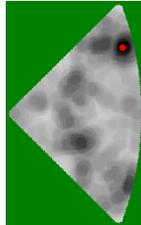
100x100 scanned at a rate of 44 fps (23 min). Step size 12 nm



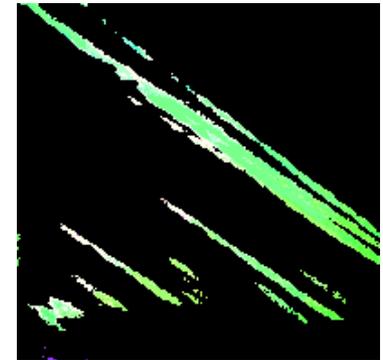
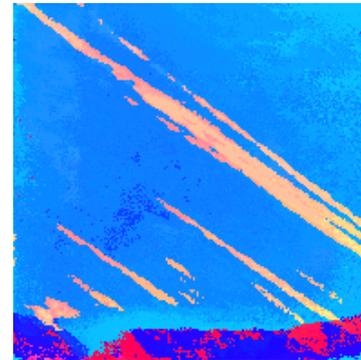
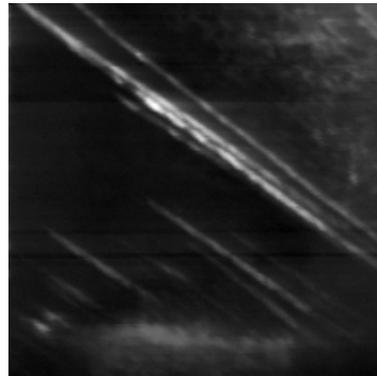
# Twin analysis in deformed steel samples



matrix reflections



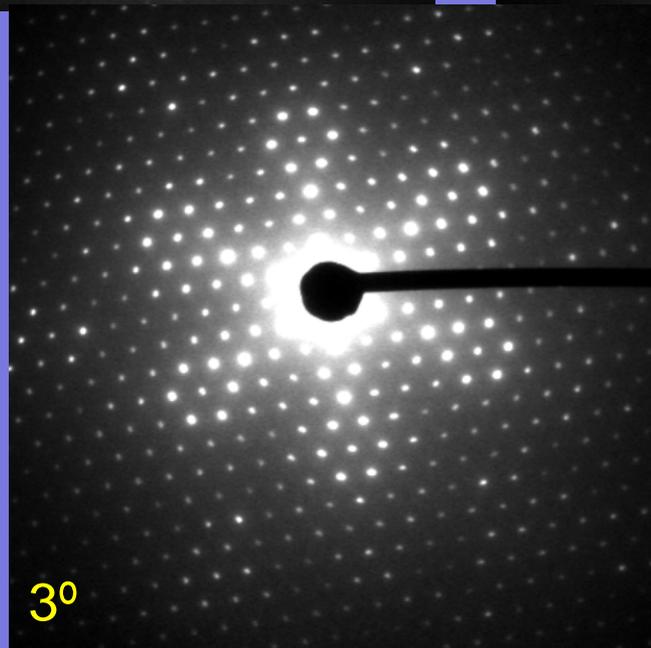
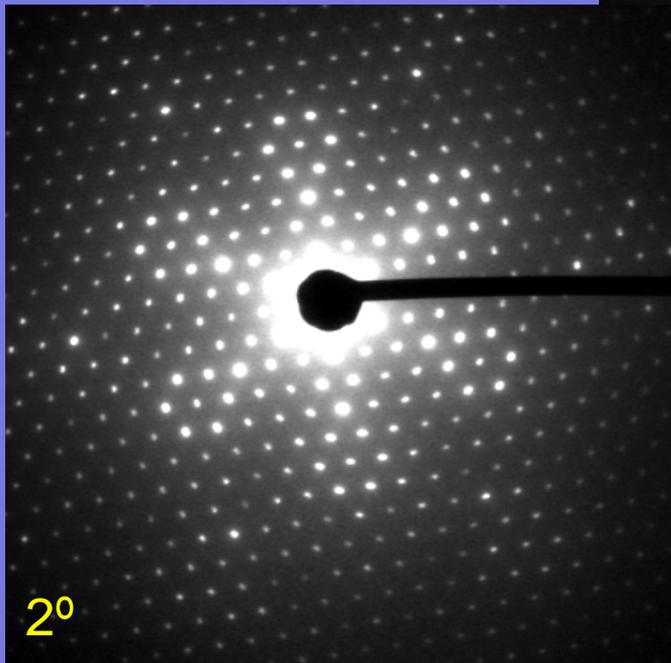
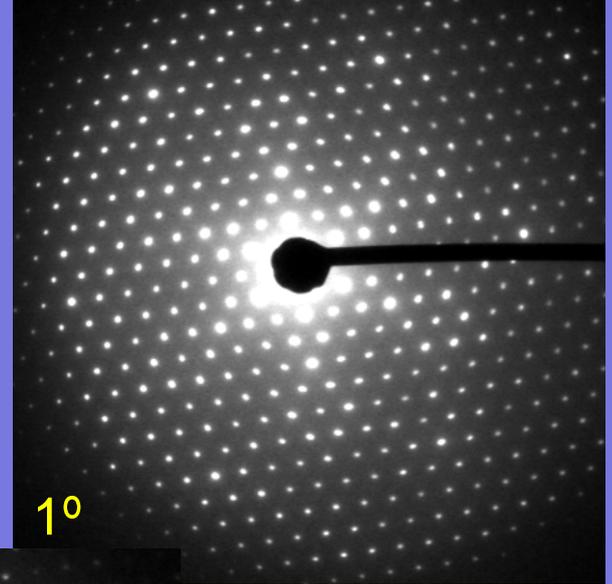
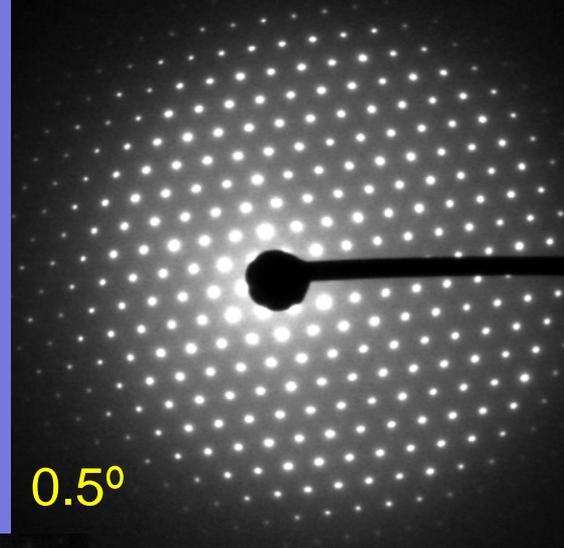
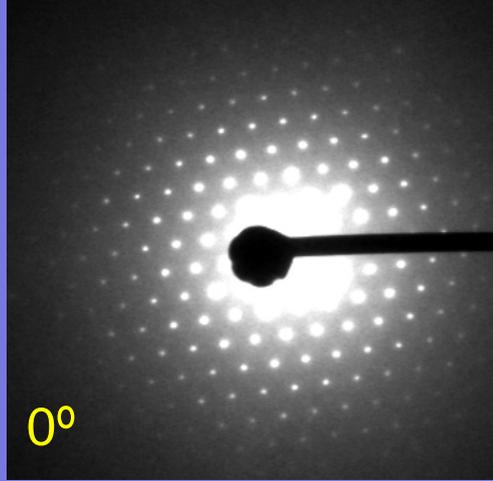
twin reflections



Virtual dark field

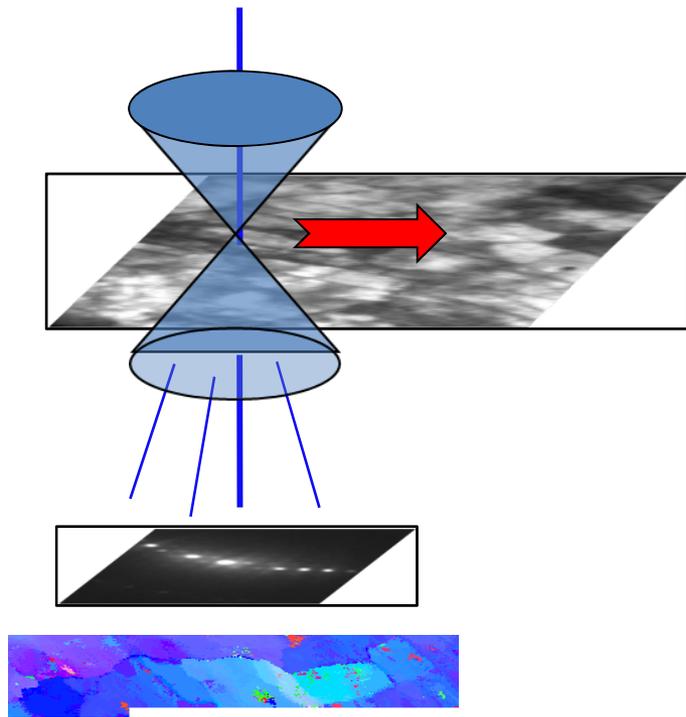
**TWIP deformed steel**





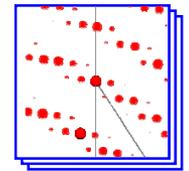
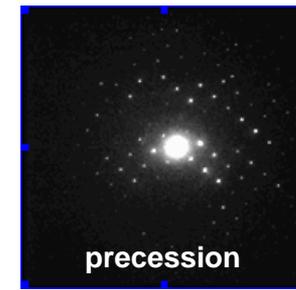
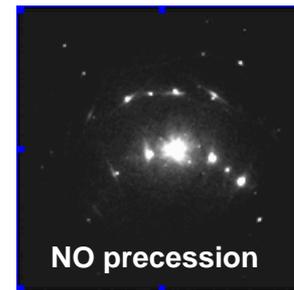
As the precession angle increases from 0° to 3°, the diffraction pattern goes to higher resolution (i.e. more diffraction spots are seen).

# ASTAR : combine scanning with precession

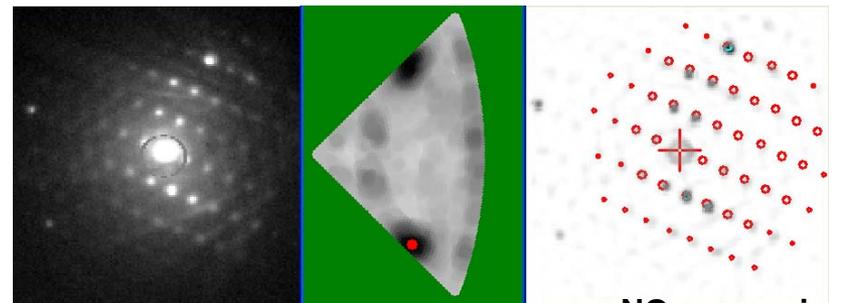


Orientation map

In this example (right) a metal particle gives wrong correlation index without precession due to presence of Kikuchi lines; after applying precession (right lower image), index gets correct value as ED quality improves and Kikuchi lines disappear

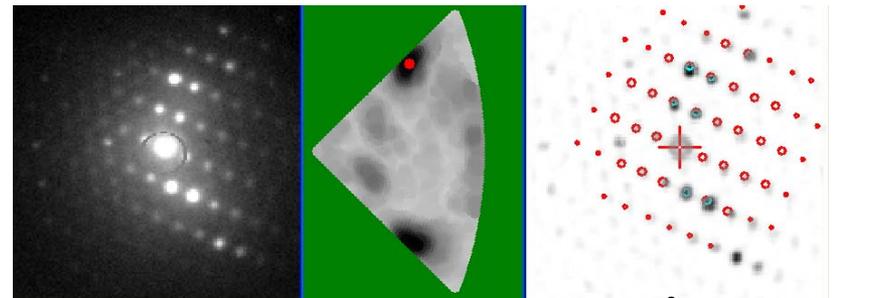


Using precession diffraction the number of ED spots observed increases ( almost double ) ; correlation index map becomes much more reliable when compared with templates



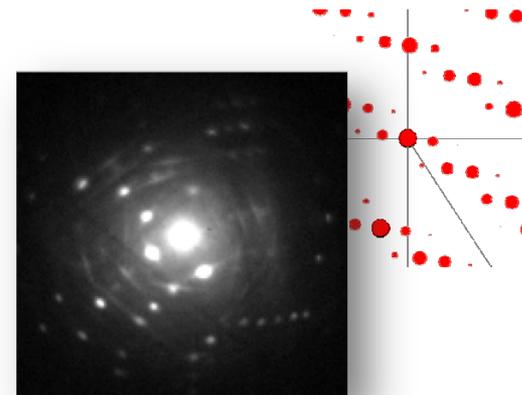
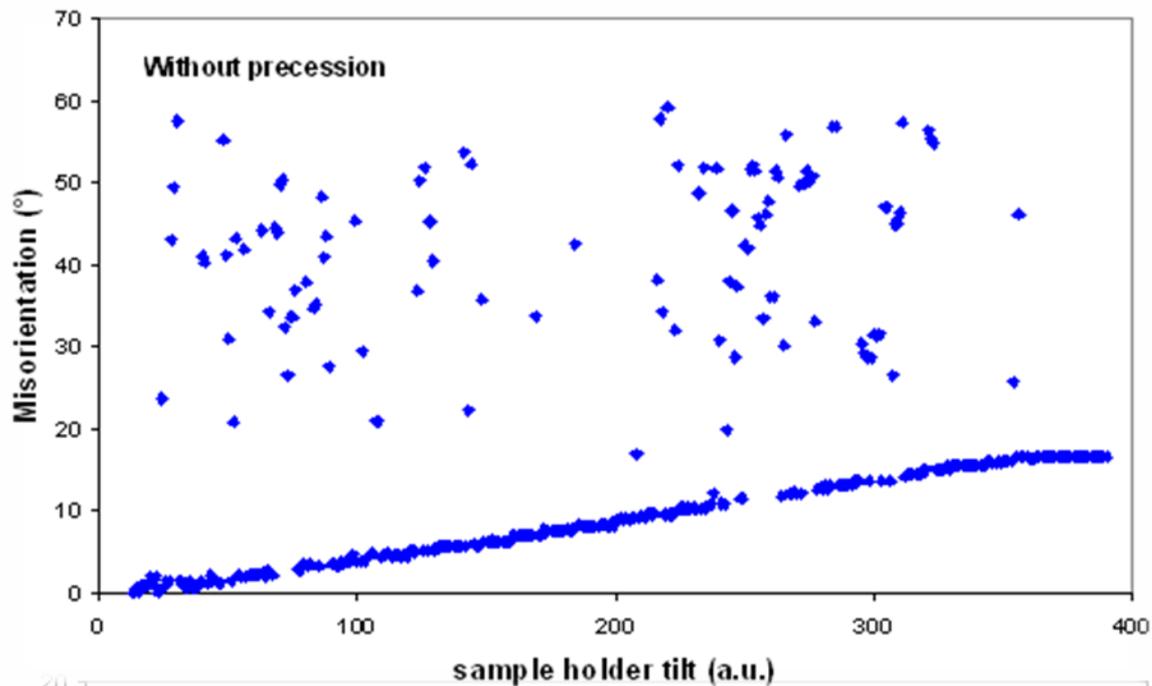
INCORRECT orientation

NO precession  
(Index 622)



CORRECT orientation

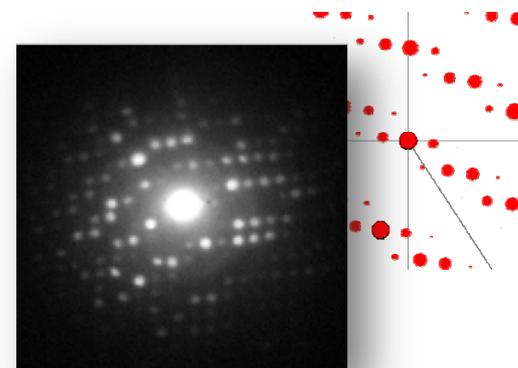
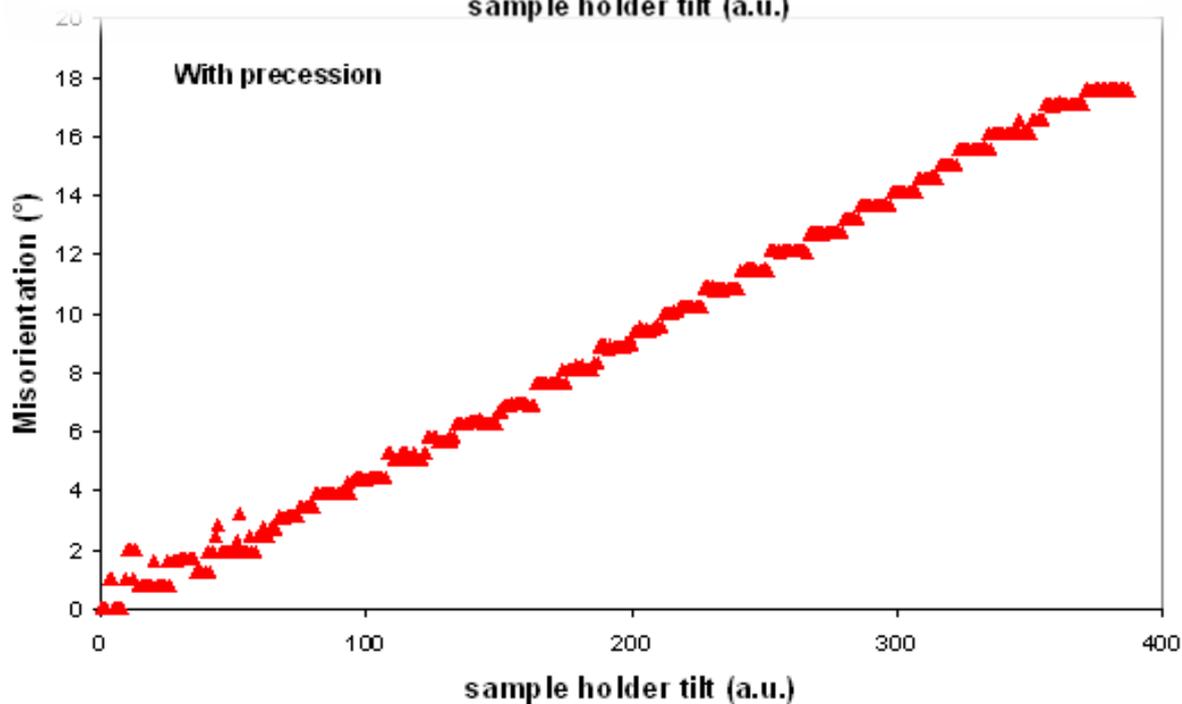
0.5° precession  
(Index 745)



**Mayenite mineral**

**Same crystal tilted 0-20°**

**400 patterns collected**

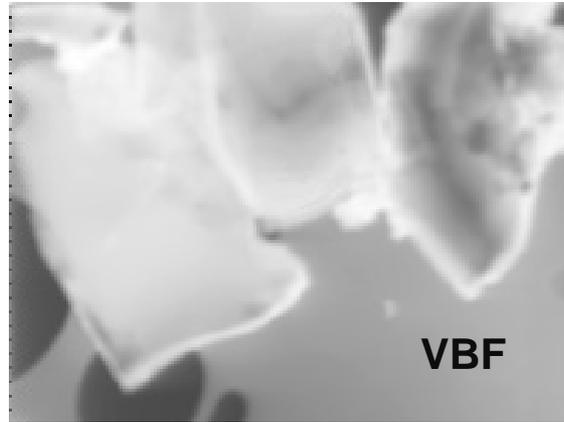
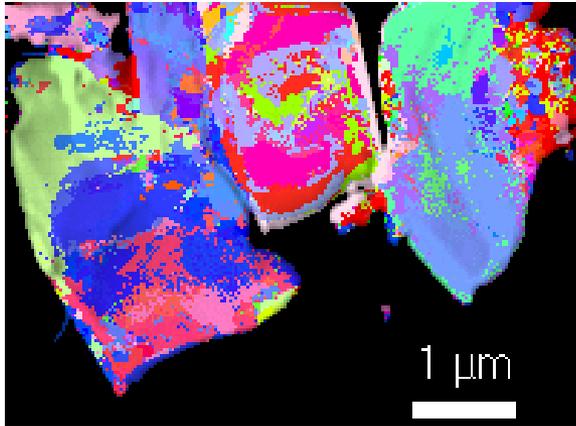


**Precession angle 0.25°**

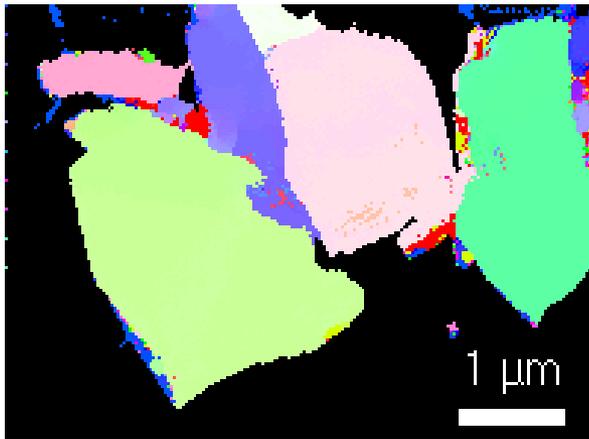


# EBSD-TEM : orientation maps with and without beam precession

orientation map, NO precession



beam scanning step 28 nm



orientation map  
precession angle 0.25°

In this example three different cubic mayenite crystals  $Ca_{12}Al_{14}O_{33}$  are analyzed with ASTAR ;

orientation map generated without precession results in inconsistent index over areas that must have uniform orientation. On the contrary, orientation maps generated with small precession angle present true uniform orientation over individual grains

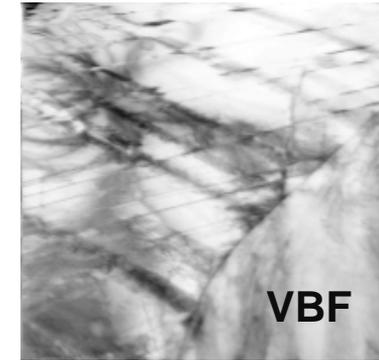
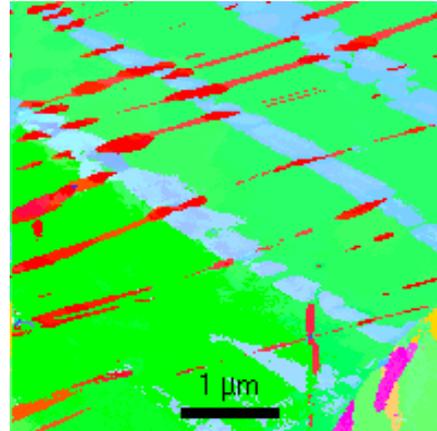
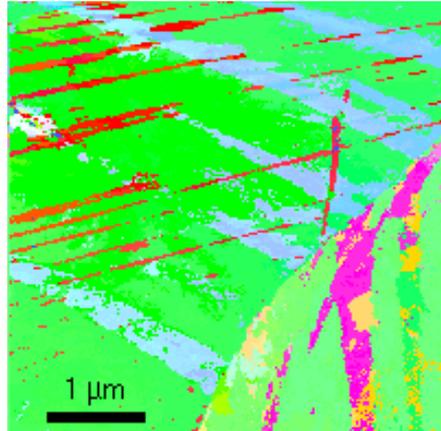
## CONCLUSION

Orientation maps are more precise with precession

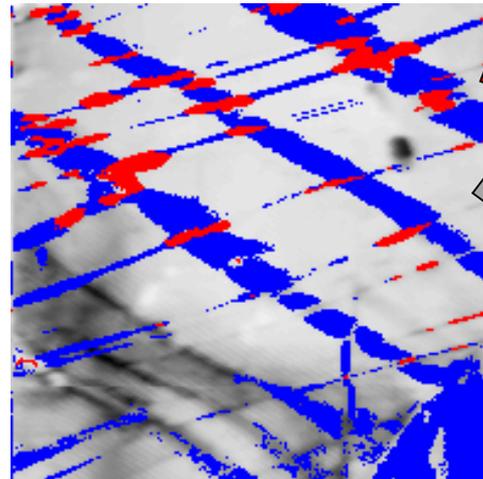
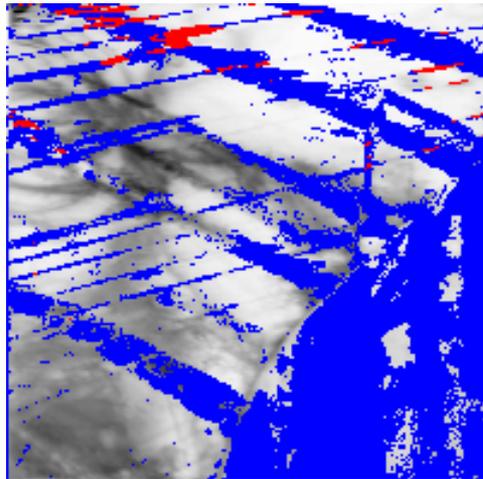
# ASTAR : Phase maps with and without precession

➔ 3 existing phases: only possible to distinguish by precession

Orientation map



crystal phase map



When stacking faults cross themselves, they produce locally  $\alpha'$  martensite structure ( $a=2.87 \text{ \AA}$ )

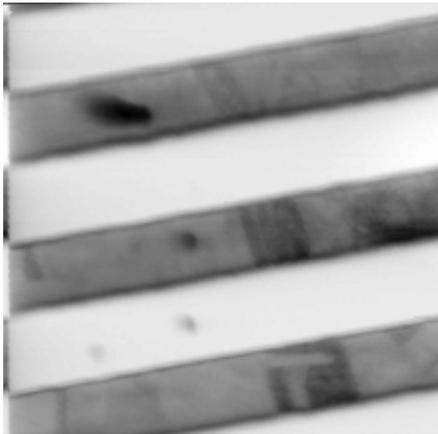
Austenitic matrix with  $\gamma$  fcc structure ( $a=3.58 \text{ \AA}$ )

Stacking faults with  $\epsilon$  hexagonal structure ( $a=2.57 \text{ \AA}$   $c=4.08 \text{ \AA}$ )

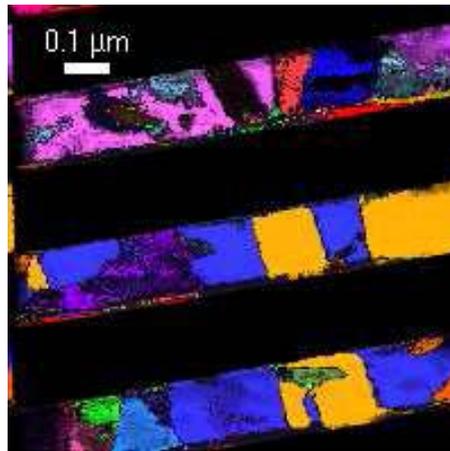
NO precession

precession  $0.4^\circ$

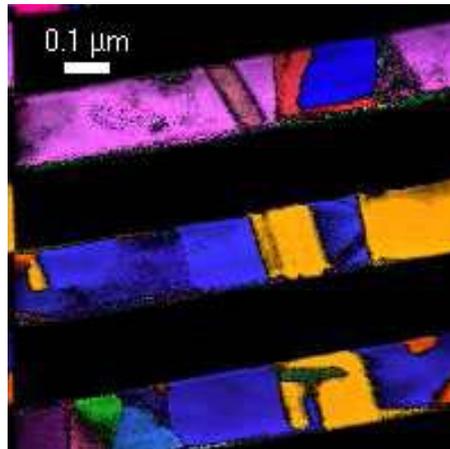
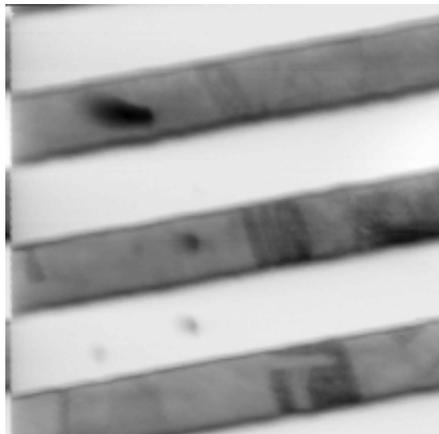
# EBSD like-TEM : copper lines ( FEG –TEM )



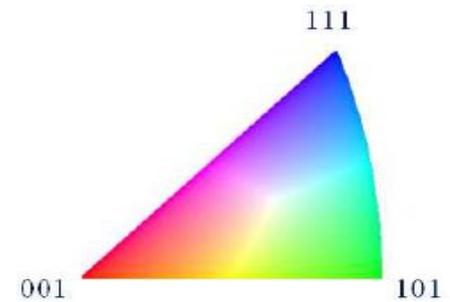
Bright field



No precession

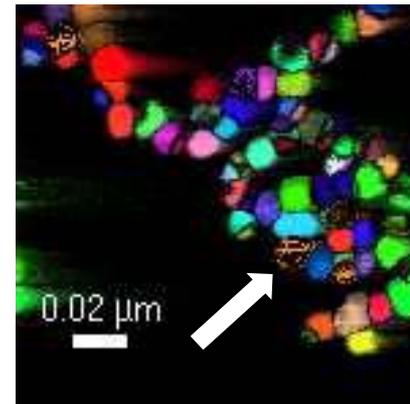
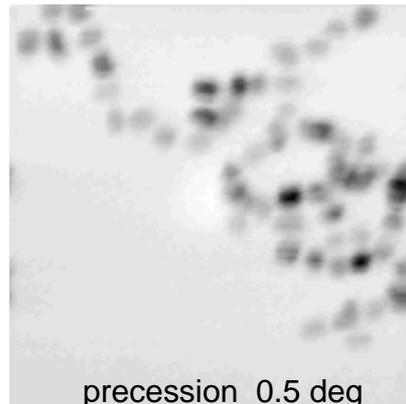
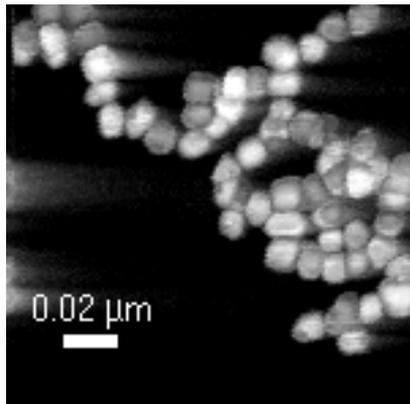
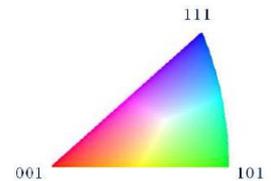
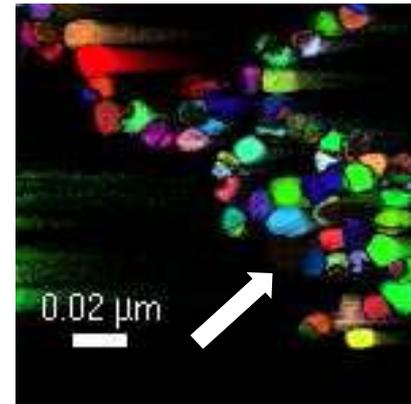
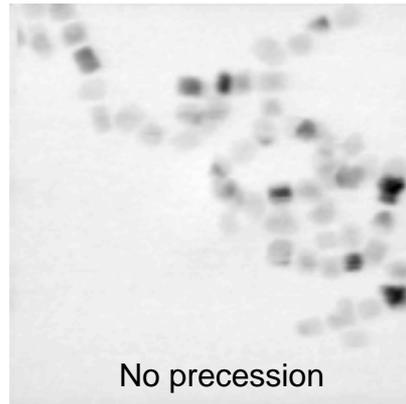
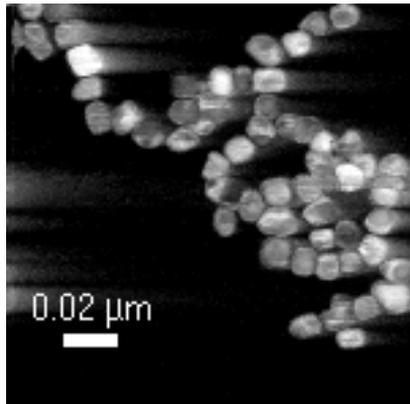


precession 0.6 deg



Jeol 2010 FEG Univ Texas -Austin , 1 nm spot size NBD mode, 200x200 pixel, **step size 9 nm**

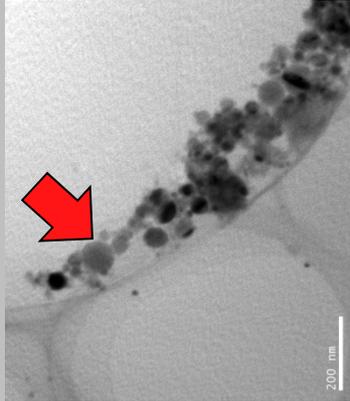
# Pt nanoparticles ( FEG -TEM )



Jeol 2010 FEG Univ Texas -Austin , **1 nm spot size NBD mode**, 150x150 pixel, **step size 1 nm**

recent unpublished results

# Nanoparticle ( 50 nm ) phase identification



cubic 8.32 Å

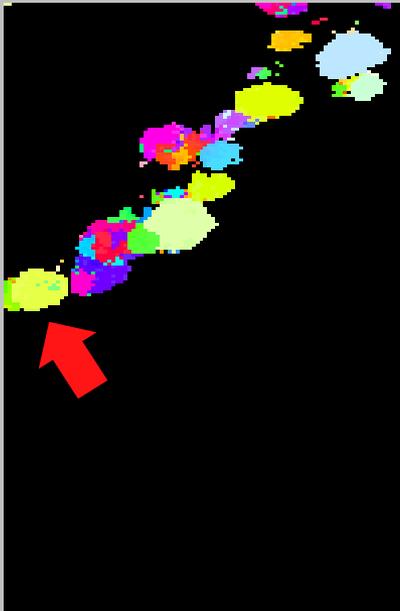
$Fd\bar{3}m$

**Magnetite or maghemite ??**

$P4_132$   $\gamma\text{-Fe}_2\text{O}_3$

$\text{Fe}_3\text{O}_4$

cubic 8.32 Å



Orientation map precession 0.3°

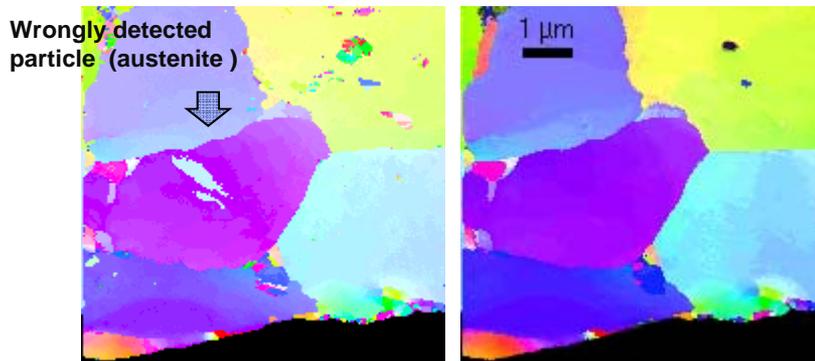
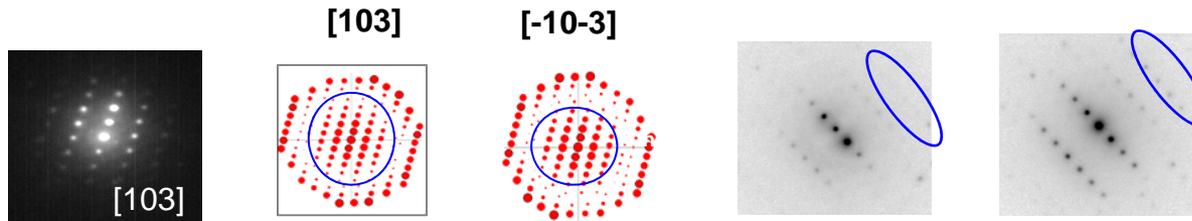


PHASE map precession 0.3°

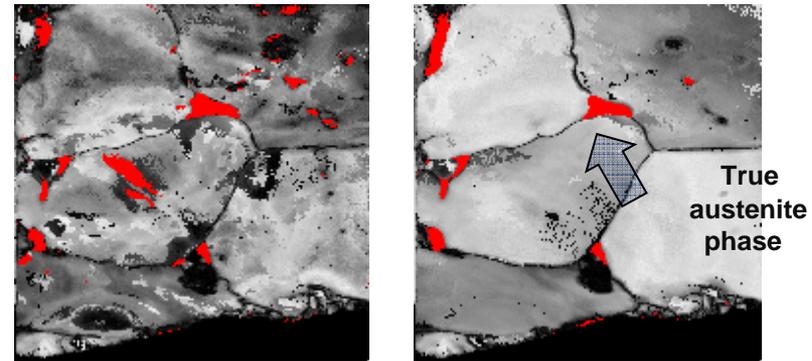
ALL Nanoparticles  
REVEALED AS  
**magnetite (RED )**

# Grain and phase boundaries: **solving 180° ambiguity** with precession

The ambiguity in the indexing of ED spot patterns arises from the fact that a particular reflection may be indexed either as (hkl) or (-h-k-l). **While this ambiguity is irrelevant for some applications, it becomes important for determination of grain and phase boundary parameters.**



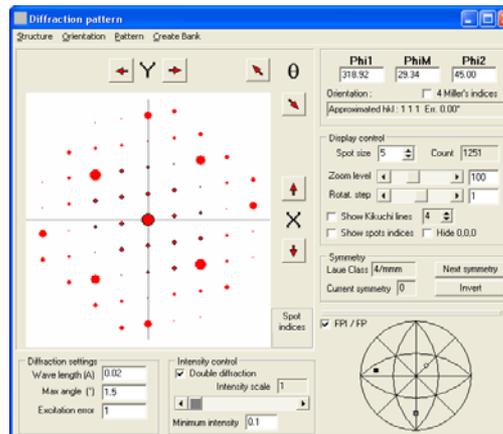
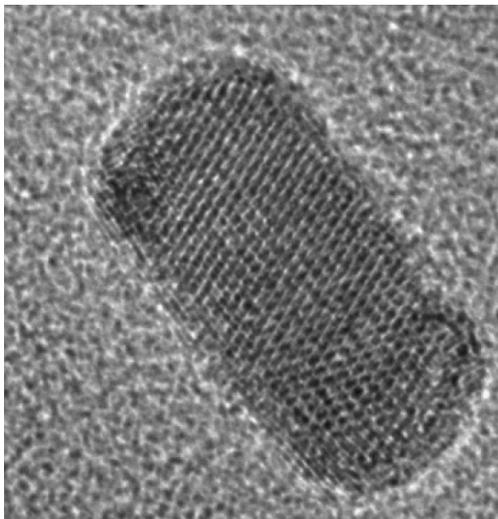
ASTAR + precession



ASTAR + precession

TRIP steel (bcc ferrite + fcc austenite) ; Philips CM120 (6 min scanning ),

# Indexing high resolution image of anatase - TiO<sub>2</sub>



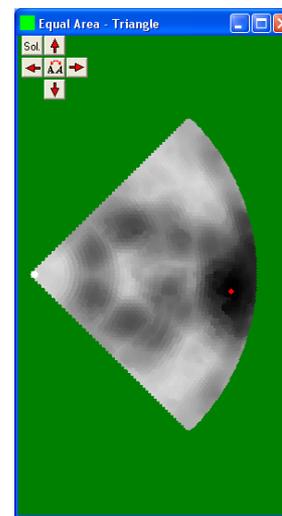
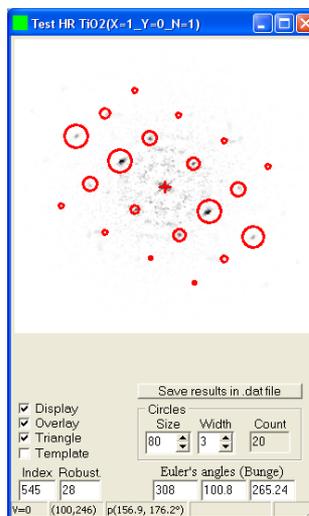
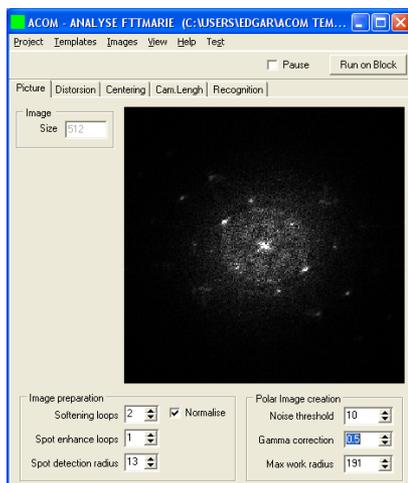
Structure name: TiO<sub>2</sub> - Anatase

Space group: 141 (41/m d)

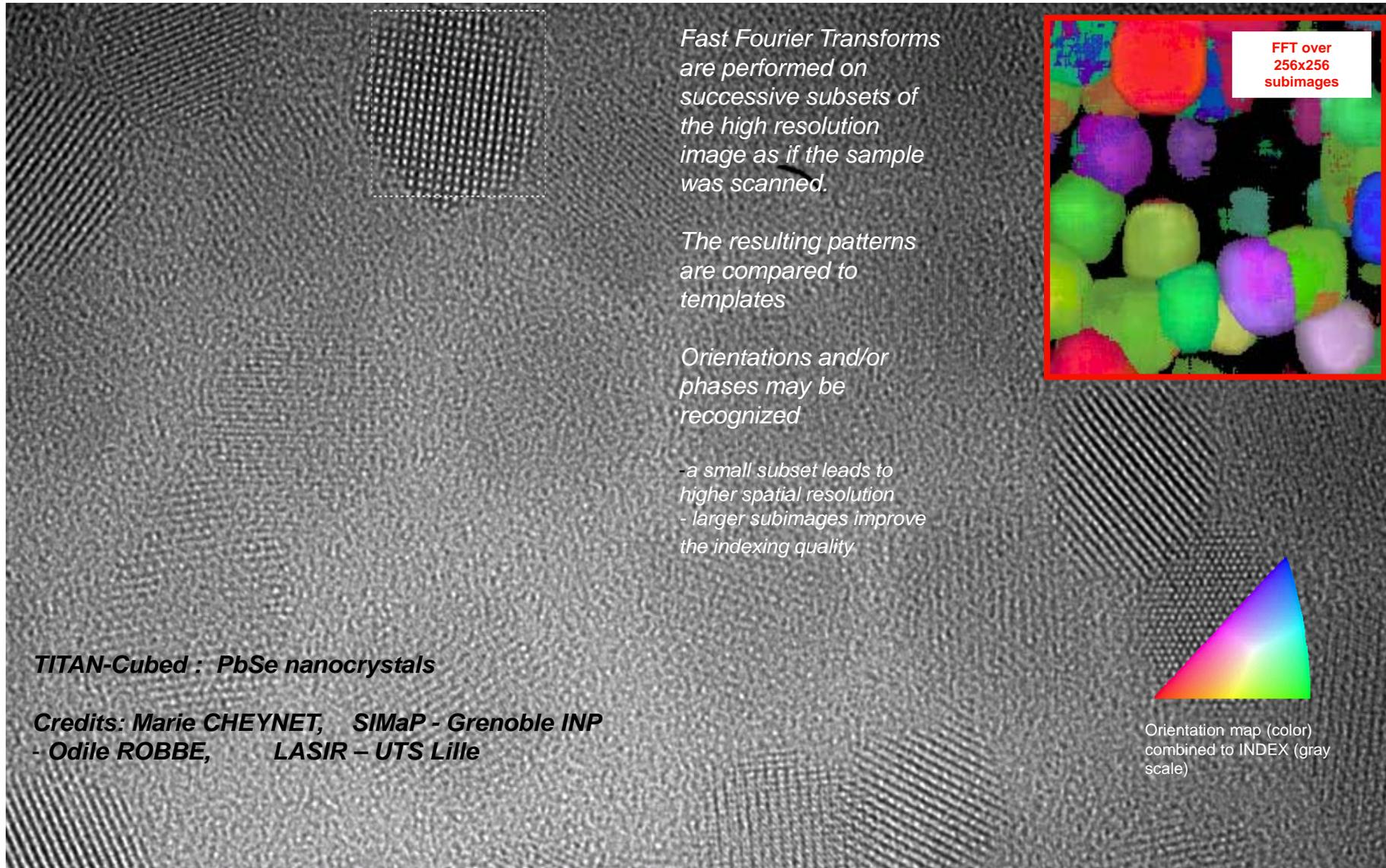
Unit cell parameters: a = 3.780, b = 3.780, c = 3.510

Elem.	Occ	F	x	y	z	Int	
1	Ti	1	22	0	0	1.000	
2	O	1	0	0	0.2081	0.360	
3	Ti	1	22	0.25	0.25	1.000	
4	Ti	1	22	0.5	0	1.000	
5	Ti	1	22	0.25	0.25	1.000	
6	Ti	1	22	0.75	0.75	1.000	
7	Ti	1	22	0.75	0.25	1.000	
8	Ti	1	22	0.5	0.5	1.000	
9	Ti	1	22	0	0.5	1.000	
10	O	1	0	0.25	0.75	0.4581	0.360
11	O	1	0	0.5	0	0.7081	0.360
12	O	1	0	0.25	0.25	0.9581	0.360
13	O	1	0	0.75	0.75	0.4581	0.360
14	O	1	0	0.75	0.25	0.9581	0.360
15	O	1	0	0	0	0.7919	0.360

it is possible to detect automatically the nanoparticle orientation: close to [100]



# ASTAR : Phase /orientation mapping HREM images

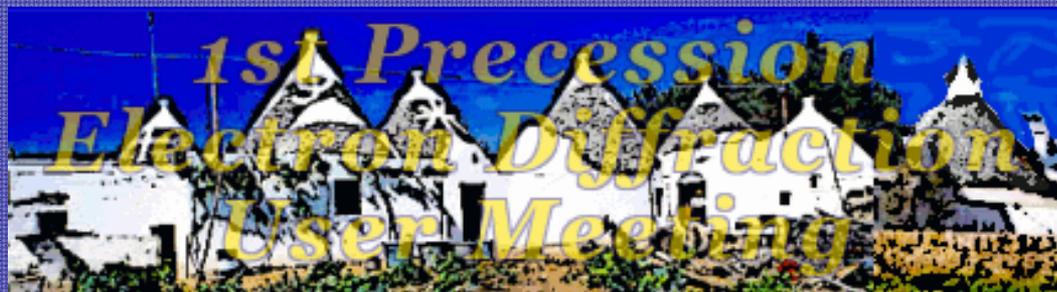


Precession adds a value to EBSD-TEM technique :

- ➔ ED patterns acquired with precession contain less dynamical effects , more spots and when compared with templates give much better correlation-reability than ED patterns without precession
- ➔ Correlation index for many reflections highly increases even at small precession angles ( eg  $0.2^\circ$  - $0.5^\circ$  )
- ➔ Orientation maps for several materials are of much better quality with precession
- ➔ Phase maps for several materials are of much better quality with precession ( much less artifacts or ambiguities )
- ➔ In orientation-phase maps “ $180^\circ$  ambiguity ” for cubic crystals can be solved using precession

# 1st precession electron diffraction user meeting

 **NanoMEGAS**  
Advanced Tools for electron diffraction



**ICCI**   
CNPq Instituto de Cristalografia

## *Martina Franca 8-9 May, 2008*

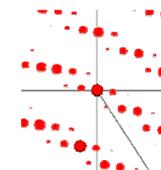
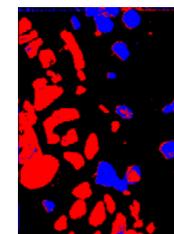
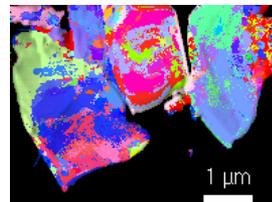
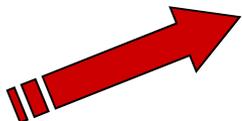
WORKSHOP ELECTRON PRECESSION EBSD-TEM

M&M Portland USA 1-5 August 2010

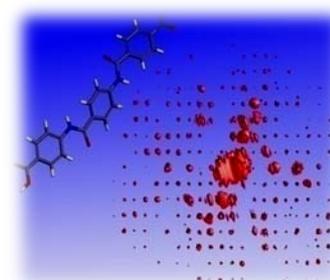
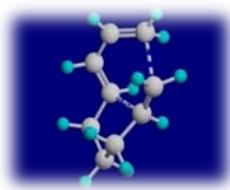
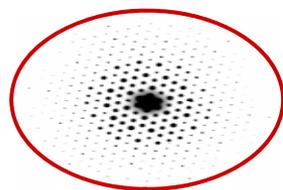
ICM17 Rio de Janeiro Brazil 22-23 September 2010

Electron Crystallography in Physical and Biological Science  
(AsCA2010 Satellite Meeting, KBSI, Daejeon) 29-30 October 2010

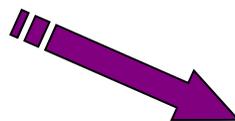
# TEM ELECTRON DIFFRACTION SOLUTIONS



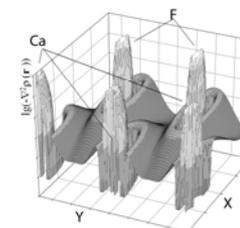
AUTOMATIC ORIENTATION / PHASE MAPPING



3D PRECESSION DIFFRACTION TOMOGRAPHY



PRECESSION UNIT  
« DigiStar »  
Basic platform



ELECTRON DIFFRACTOMETER-